

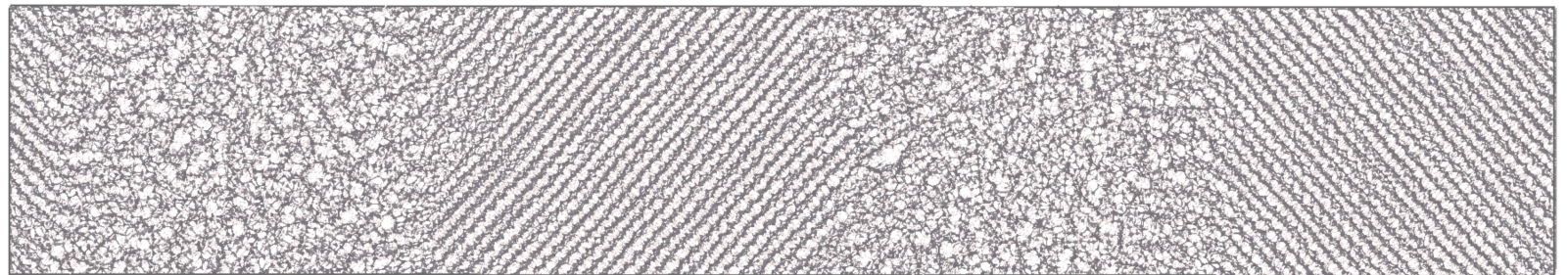
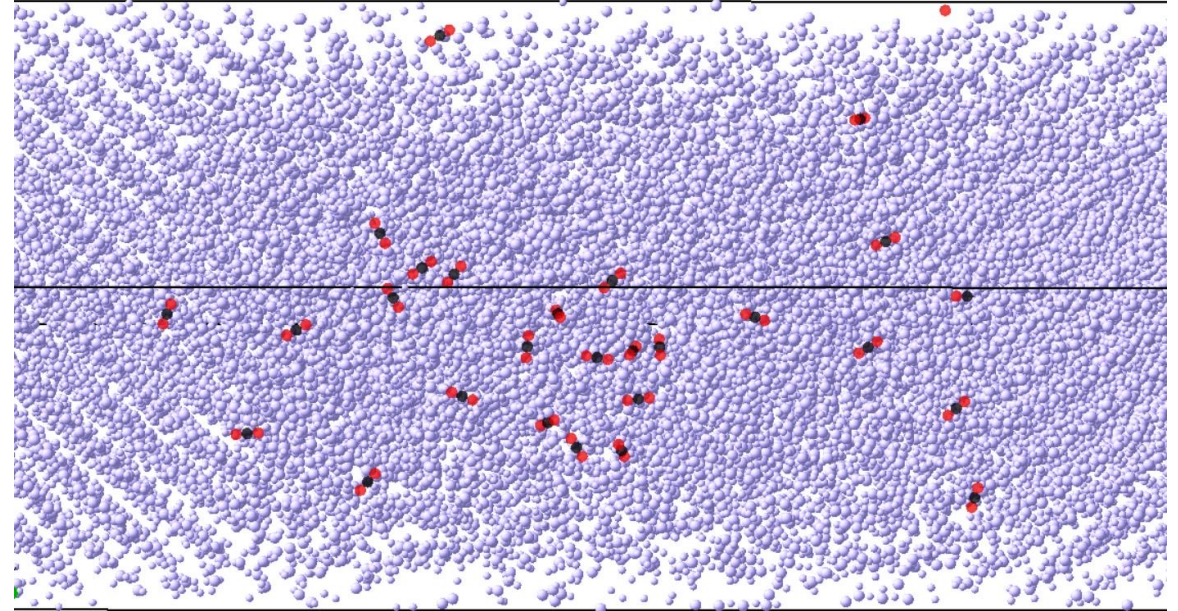


materials design

Sorption and Diffusion of Small Gas Molecules in Semicrystalline polymers: A Molecular-Scale Investigation

Speaker: Dr. Boris Belin

March 5th-7th, 2024



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Webinar Speakers

*Katherine
Hollingsworth*

*Dr. Dave
Rigby*

*Dr. Marianna
Yiannourakou*

*Presenter:
Dr. Boris Belin*

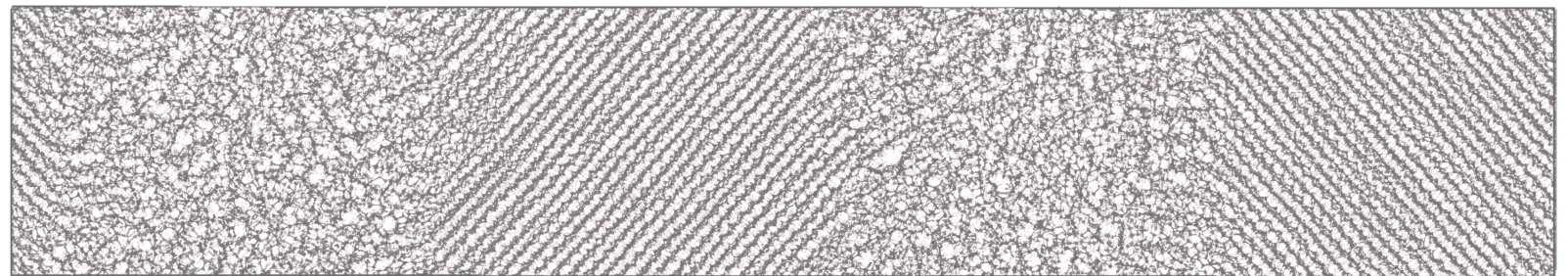
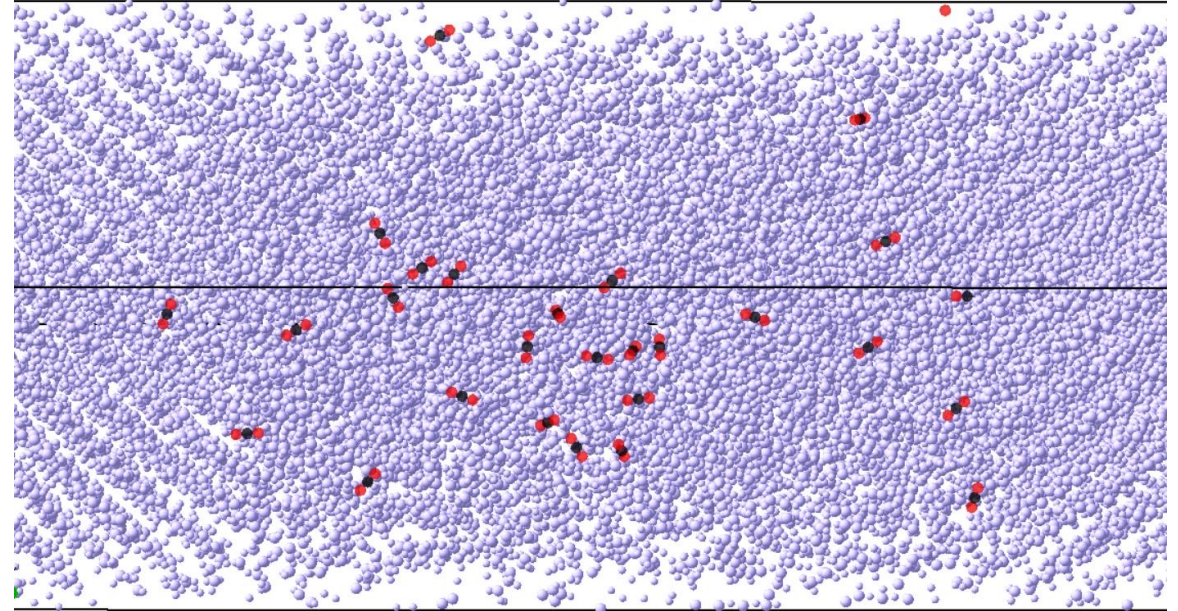


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Sorption and Diffusion of Small Gas Molecules in Semicrystalline polymers: A Molecular-Scale Investigation

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Outline

- **Building semicrystalline polymer structures (reminder of the last webinar)**

- **Sorption of CH₄ and CO₂**
 - Methods
 - Results
 - Comparison with experiments with Michael's theory of sorption in S-C

- **Diffusion of CH₄ and CO₂**
 - Methods
 - Results

Semicrystalline polymers in industry

Ubiquitous !

Many properties of interest accessible with molecular/atomistic simulation:

- **Mechanical** (molecular dynamics).
- **(non)-permeation** (Monte-Carlo, molecular dynamics).
- **Dielectric/electronic** (density functional theory).

Non-exhaustive industrial uses:

- Replace metals for mechanical properties.
- Protect food and tanks or pipes from corrosion.
- Electric insulator for wires or semiconductor in O-Led TV.

POLYETHYLENE EXAMPLES





Morphology

Spherulite and shish-kebab

- Semicrystalline polymer composed of amorphous and crystalline phase.

X-ray powder HDPE

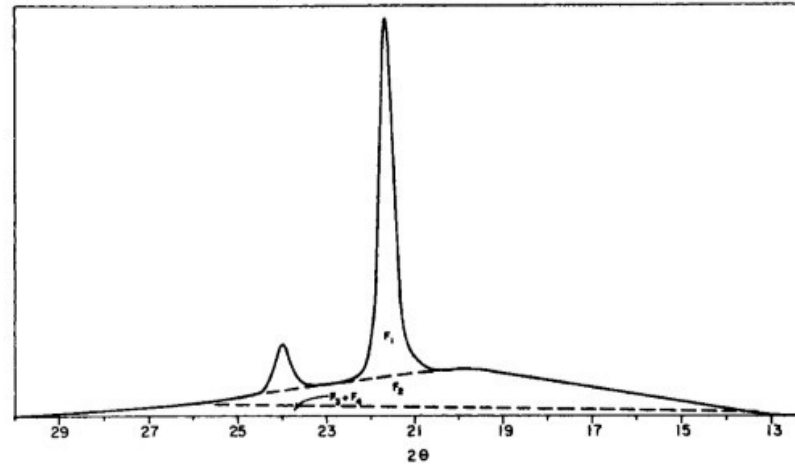
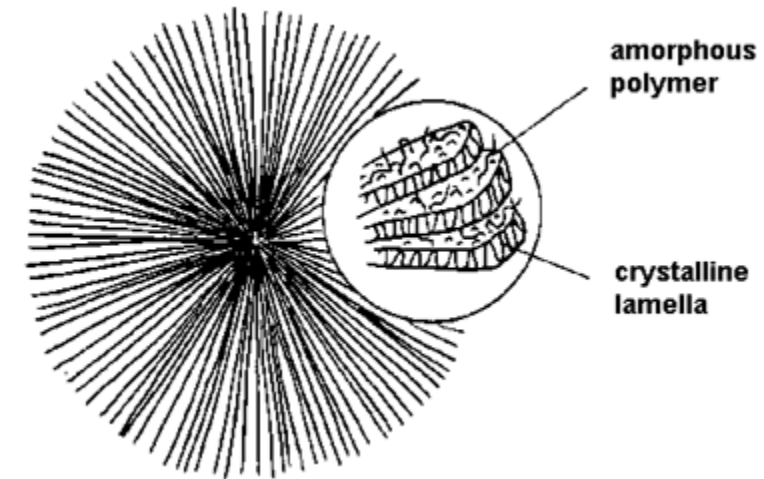


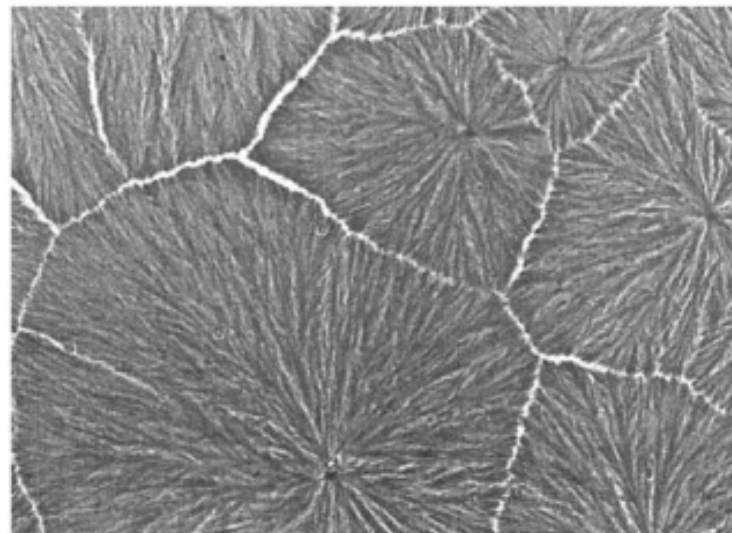
Fig. 1. X-ray diffraction pattern for linear polyethylene at room temperature. $M_w = 7 \times 10^6$; density = 0.9309.

- At the microscale: spherulite or shish-kebab (films, solidified under strain).

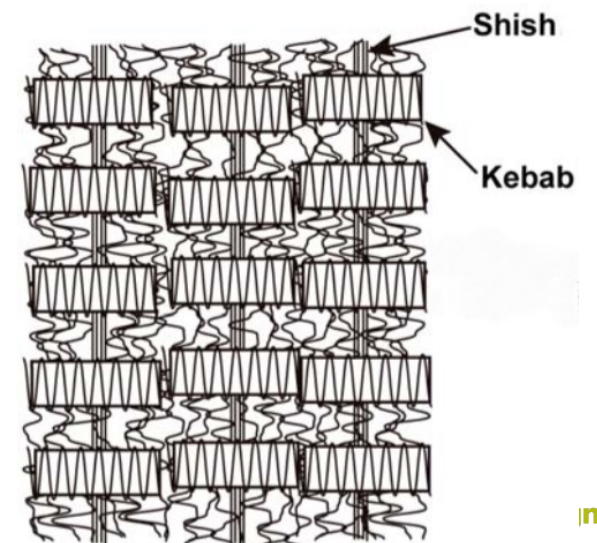
Spherulite



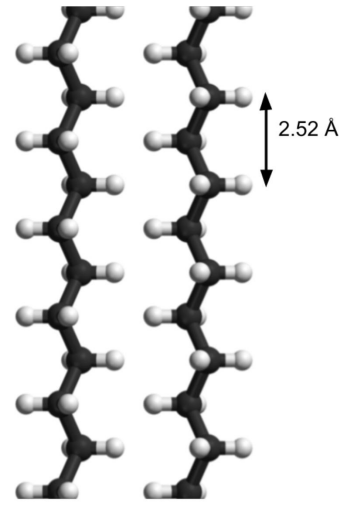
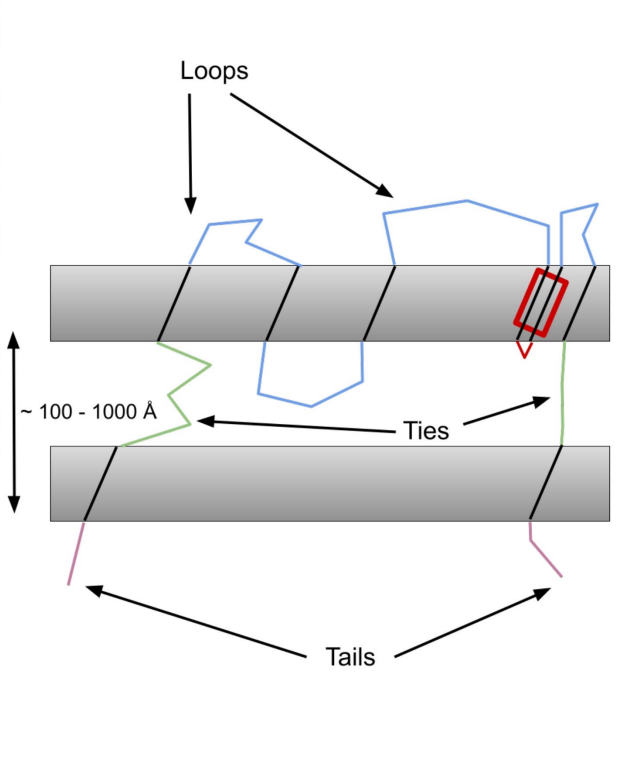
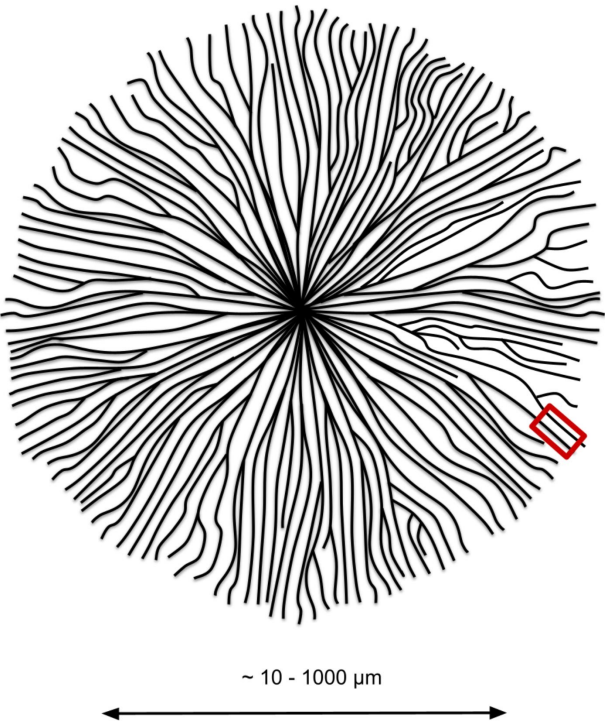
Spherulite under microscopy



Shish Kebab

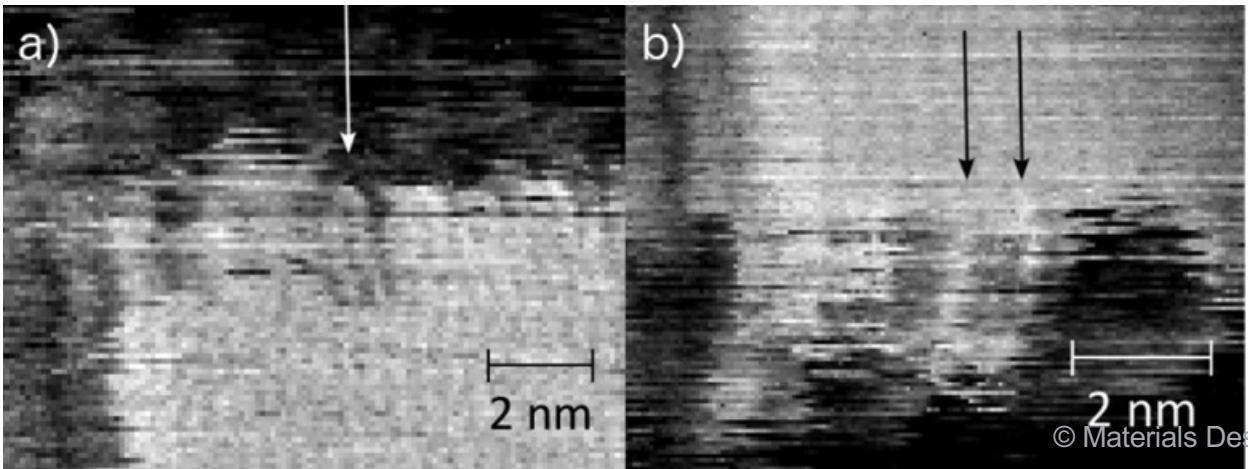


Morphological characterization



Characterization

- Density, long period, degree of crystallinity (χ), molecular weight.
- Tilt angle.
- Characterization of the amorphous section (tie chains, loops, tails).
- Characterization of the interphase (perfect foldings or not).



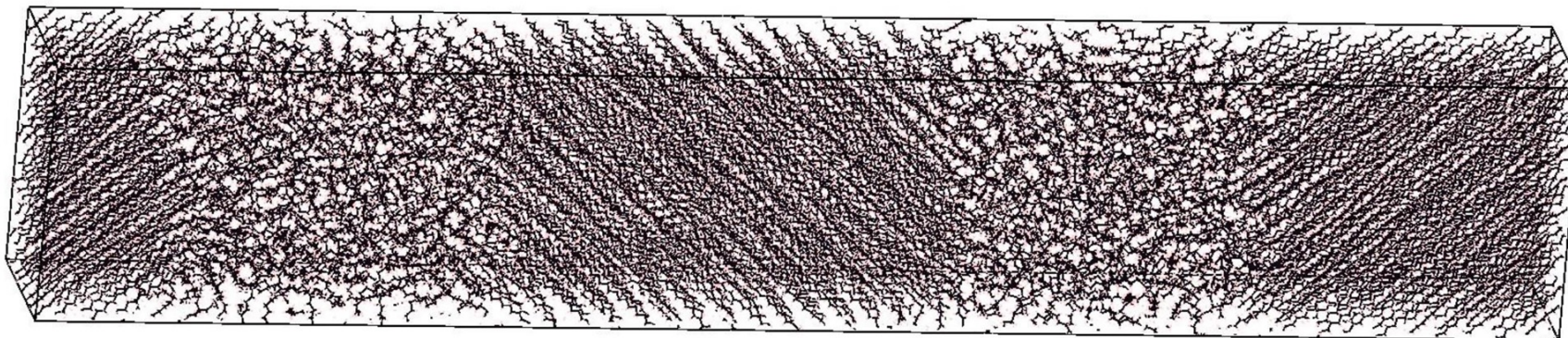
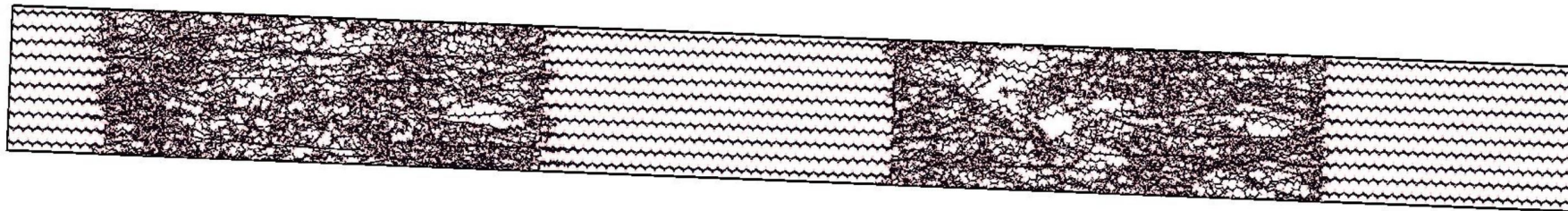
AFM study of the interphase.
R. C. Savage, N. Mullin, and J. K. Hobbs, *Macromolecules* (2015)



Building structure reminder

Overview of the method (3) : Realistic HDPE system.

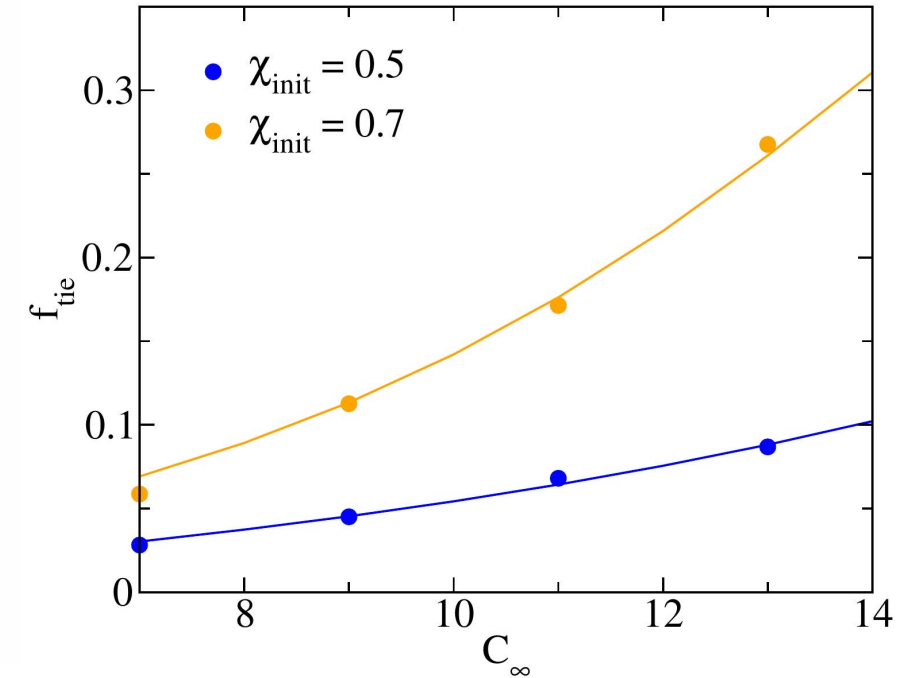
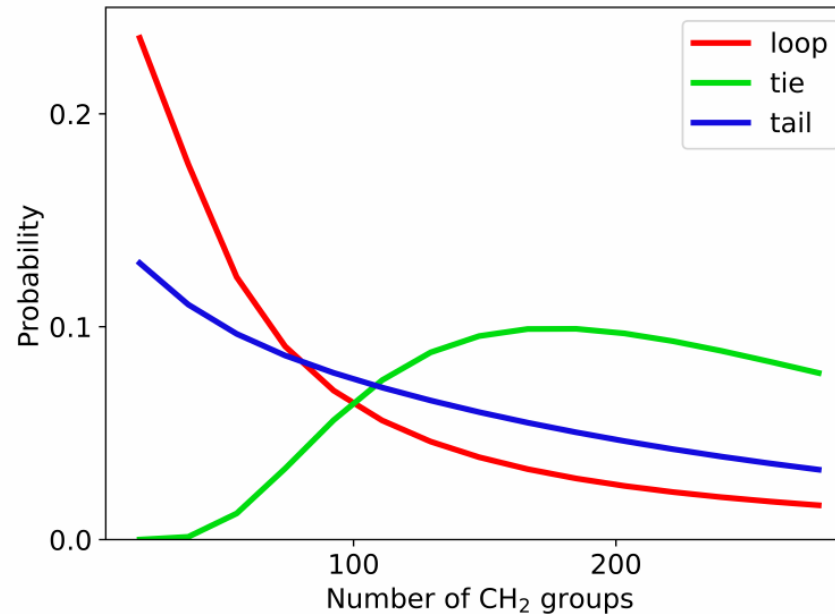
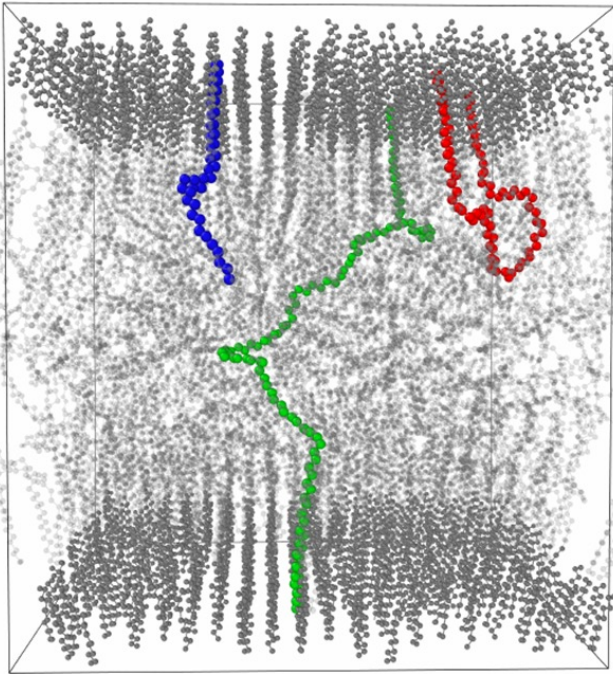
- **Hypothesis** : We know the distribution of size and type of the amorphous chain sections (tie chains, loops, tails), from experimental data or from a statistical polymer physics theory.
- With the **MedeA Amorphous Builder**, build an amorphous layer correctly labelled (depending of types). Build a supercell layer of the crystal with labelled chain ends.
- Connection (**MedeA Thermoset Builder**) and long relaxation (100 ns, ff TraPPE-UA), (AFTER Connection only two long chain left with realistic molecular weight : $\sim 10\,000$ CH_2 groups, Mw 140 000 g/mol (30 000 to 6 000 000 g/mol in exp.).



Fraction and size distribution of **loops**, **ties** and **tails**.

THEORY I USED : Theory of statistics of ties, loops, and tails in semicrystalline polymers **FREE**

Sabin Adhikari  ; Murugappan Muthukumar 



In my work, I used a new theoretical freely jointed chain statistics, which allowed me to control the fraction of tie chains f_{tie} (end others as a result), controlling the conversion between real chain and ideal chain with the Kuhn length.

It also gave the probability distribution of amorphous chain section sizes.

Characterization : Density and degree of crystallinity

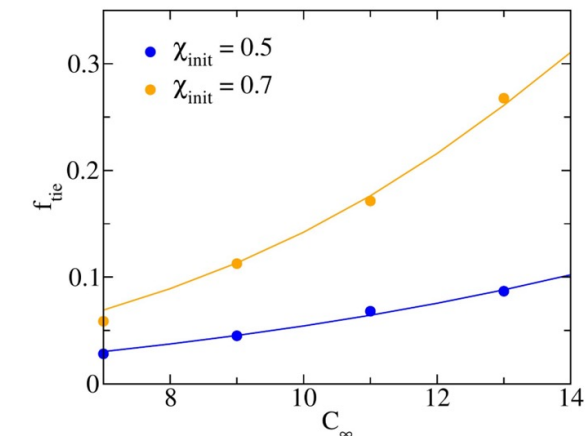
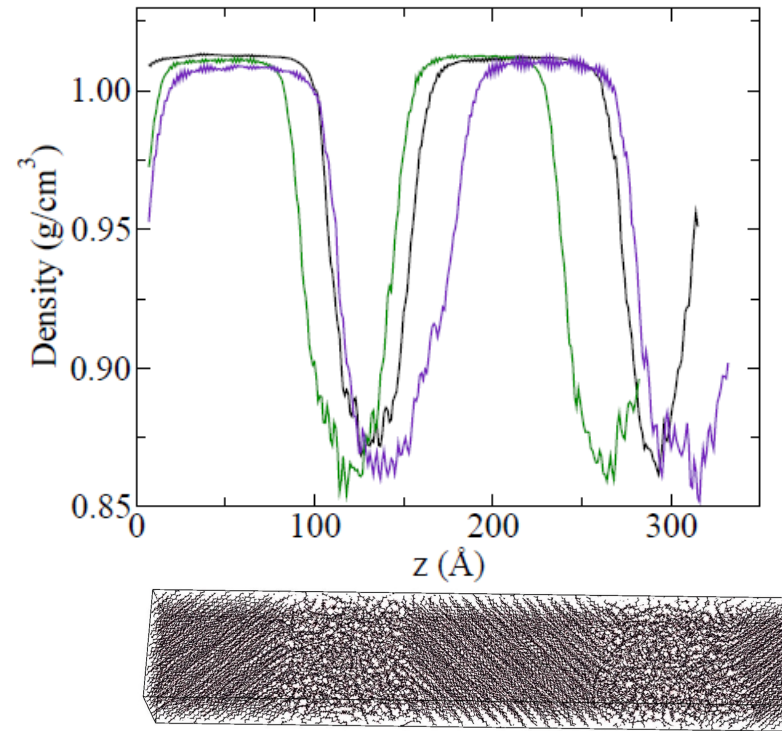
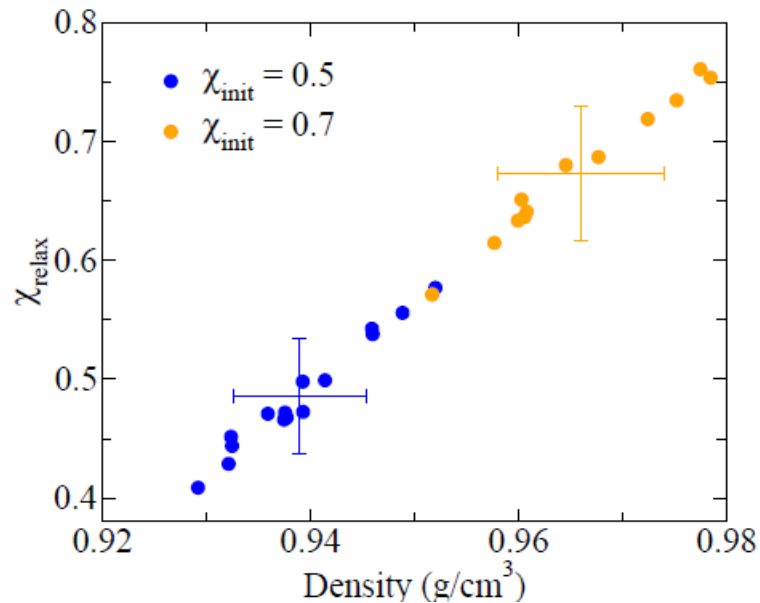
χ_{init}	0.5	0.7
$C_\infty = 7$	3	3
$C_\infty = 9$	6	3
$C_\infty = 11$	3	3
$C_\infty = 13$	3	3

χ_{init}	0.5	0.7
$\bar{\chi}_{relax}$	0.49	0.67
$\sigma_{\chi_{relax}}$	0.05	0.06
$\bar{\rho}_{relax} \text{ g/cm}^3$	0.939	0.966
$\sigma_{\rho_{relax}}$	0.008	0.006

27 structures varying the initial degree of crystallinity χ_{init} and f_{tie} ($\leftrightarrow C_\infty$).

Spread of degree of crystallinity and density.

Tilt angle apparition (in agreement with exp.)

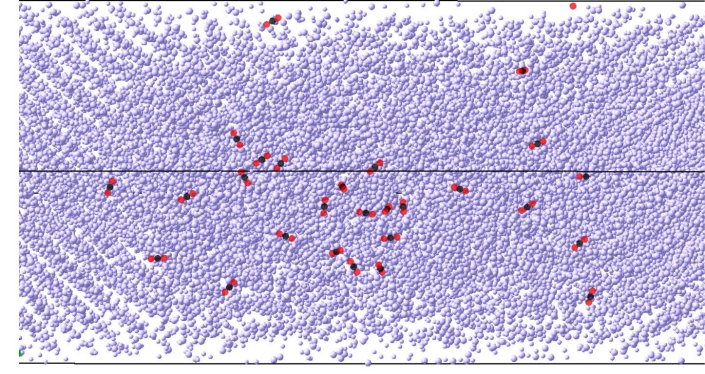
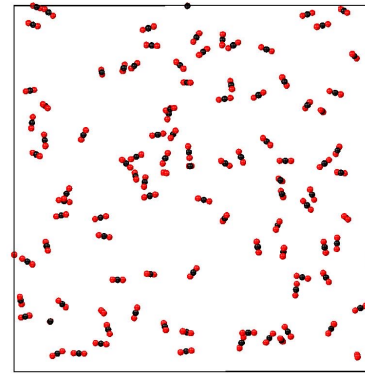


Sorption computations with CH₄ and CO₂

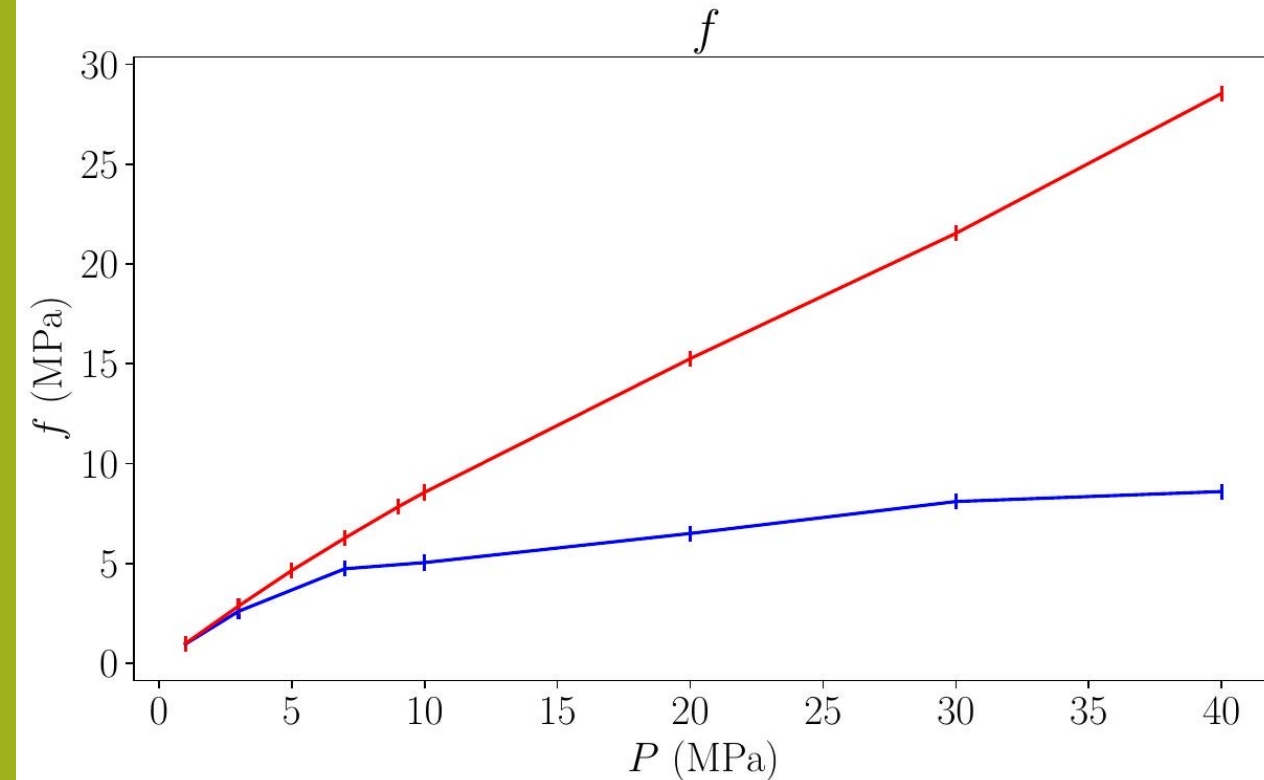
Fugacity computations

$$(\mu_{CO_2}(P, T))_{bath} = \mu = (\mu_{CO_2}(P, T))_{poly}$$

$$\mu - \mu^0 = RT \ln f/P^0$$



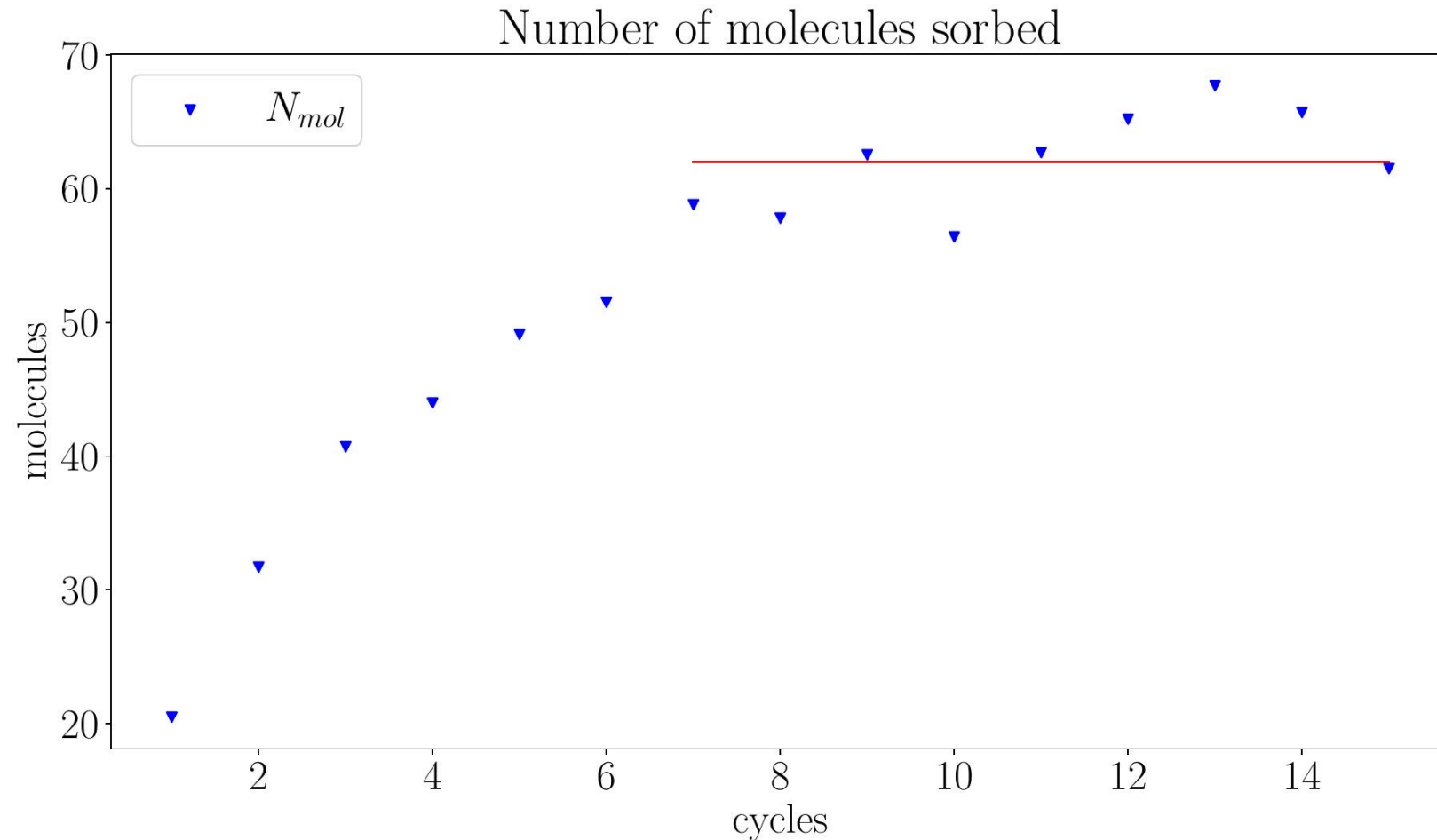
Chemical equilibrium



- Goal: We want to do sorption computations in: $\mu_i VT$ ($f_i VT$), the grand canonical ensemble where μ_i is the chemical potential.
- Chemical equilibrium: $(\mu_i)_{poly} = (\mu_i)_{bath}$.
- Widom insertion tests at P and T=300 K.

MC/MD cycles method

- **MC-MD cycle:** iteration of Monte Carlo (μ_iVT) + MD in NPT until convergence.
- **Monte Carlo run (grand canonical):** frozen polymer
- **Molecular dynamics run (NPT):** account for the chain mobility and swelling.
- **Iteration until convergence**



Velioglu, S. et al. Journal of Membrane Science (2012)

Abedini, A. et al. Langmuir (2017)

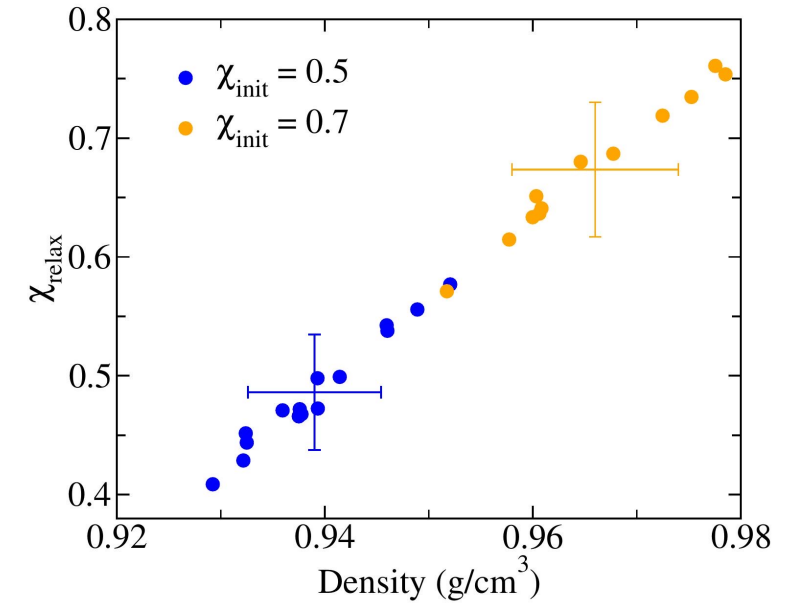
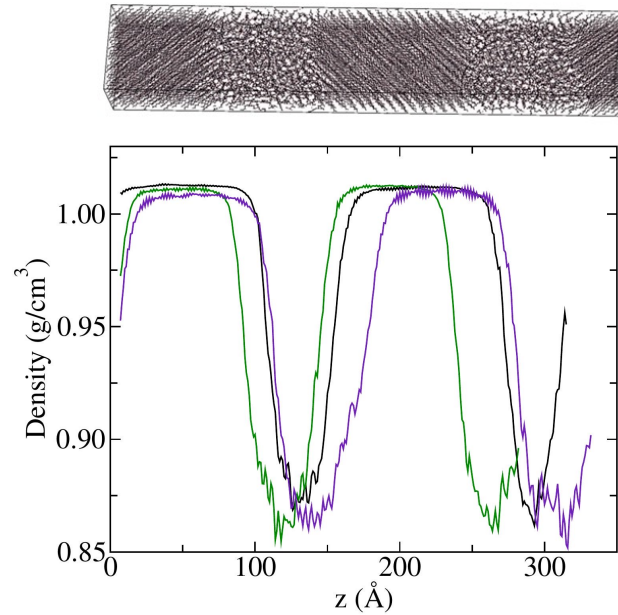
Velioglu, S. et al. Journal of Membrane Science (2018)

Kupgan, G. et al. Journal of Membrane Science (2018),

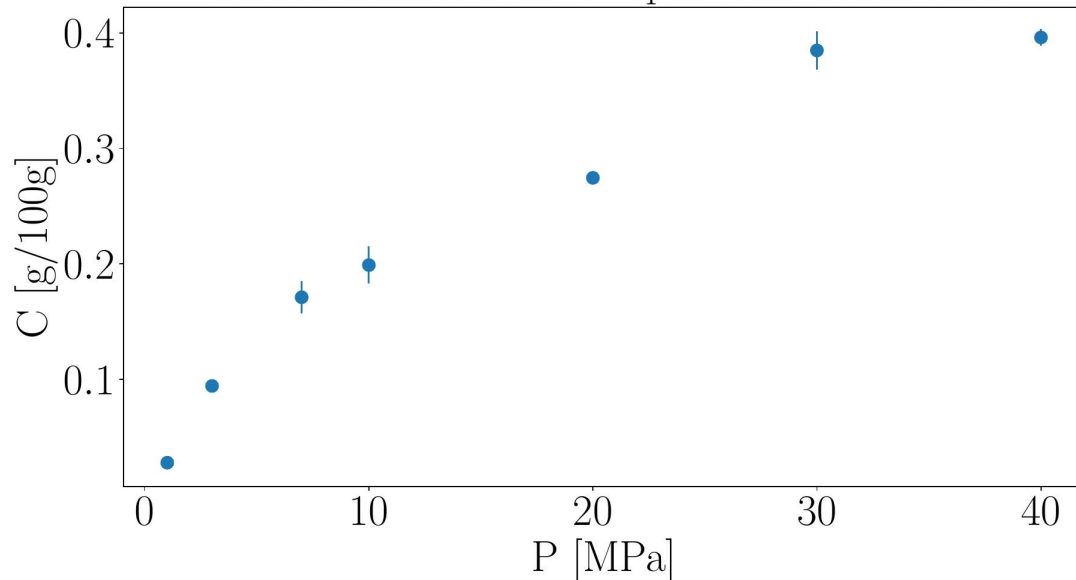
Balcik, M. & Ahunbay, M. G. Journal of Membrane Science (2018)

Michaels' Law

- Michaels' Law, to compare results : $C_a = C / \Phi_a$ (Φ_a is the volumic amorphous fraction).
- $\rho_a = 0.86 \text{ g.cm}^{-3}$ $\rho_c = 1 \text{ g.cm}^{-3}$

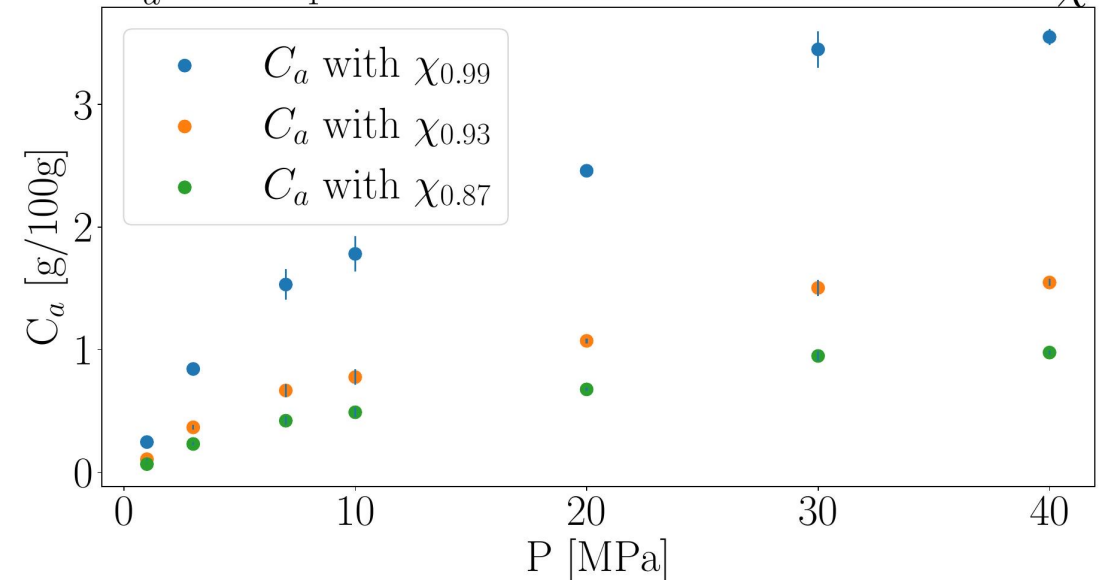


Concentration of CH₄ in one structure



Example of absorption isotherm

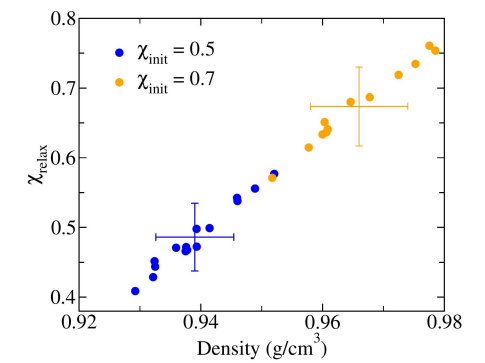
C_a of CH₄ with three different definitions for χ



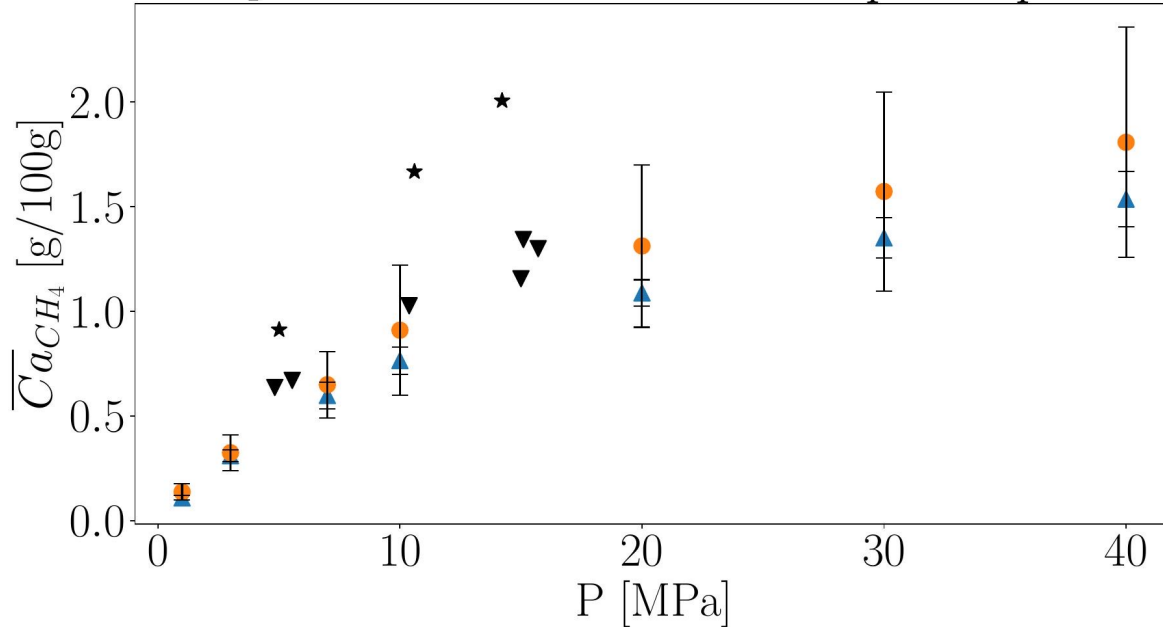
CH₄ absorption isotherm in the amorphous

Results averaged over all the structures

Michaels' law is verified, and predictions are in agreement with experiments.
No influence of tie chains or entanglements.



CH₄ concentration in the amorphous phase



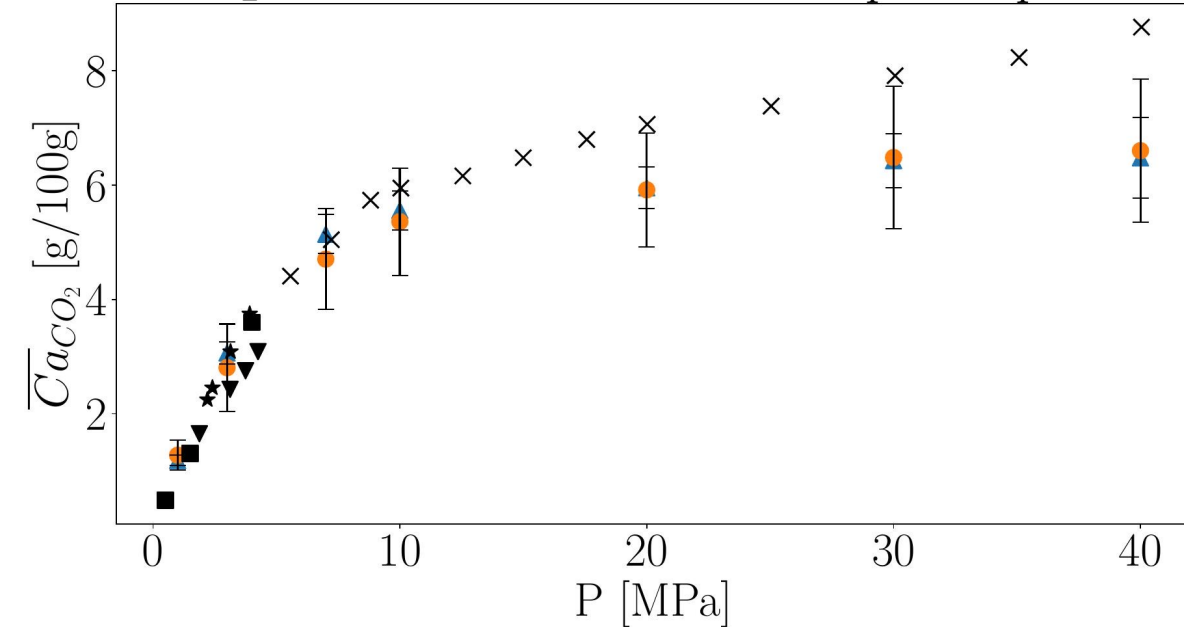
Concentration of CH₄ in the amorphous (blue triangles $\chi_{init} = 50\%$) and (orange dots $\chi_{init} = 70\%$) at T=300 K. Von Solms' measurements (\blacktriangledown , T=305 K) and (\star , T=298 K).

Von Solms, N. et al. Journal of Applied Polymer Science (2004)

Flaconeche, B. et al. Oil & Gas Science and Technology - Revue d'IFP Energies nouvelles (2001)

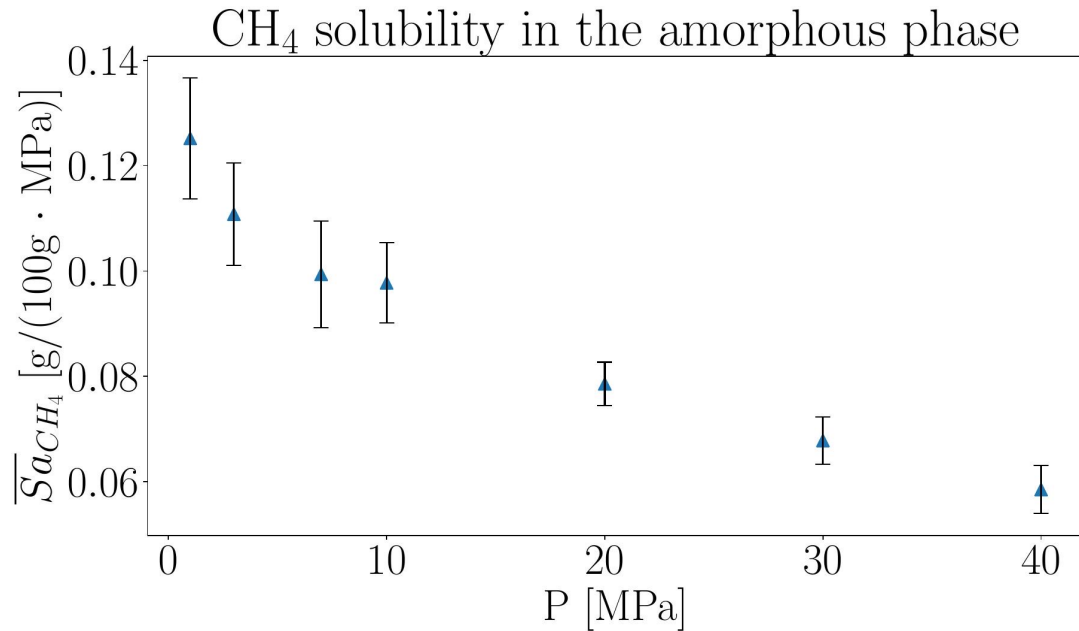
Hu, T. Permeation of High Pressure CO₂ in Semicrystalline Polymers. Phd Thesis. (Imperial College London, 2021)

CO₂ concentration in the amorphous phase



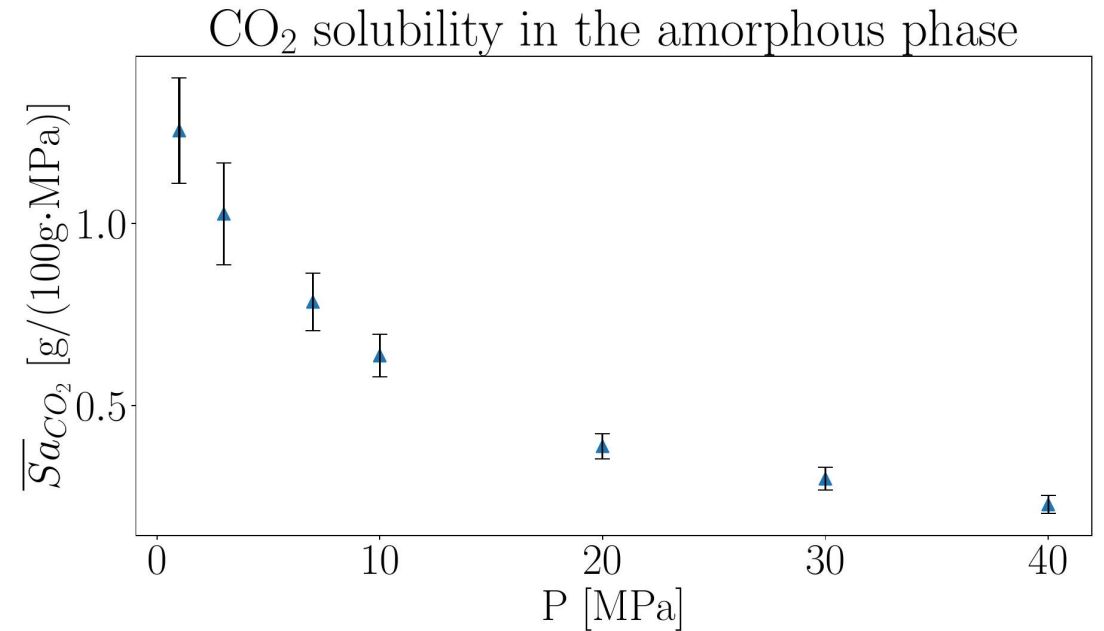
Concentration of CO₂ in the amorphous fraction (blue triangles $\chi_{init} = 50\%$) and (orange dots $\chi_{init} = 70\%$) at T=300 K. Von Solms' measurements (\blacktriangledown , T=305 K) and (\star , T=298 K). Flaconeche et al. squares (\blacksquare , T=298 K), and Hu cross (\times , T=298 K).

Solubility coefficients



Average CH₄ sorption coefficient in the amorphous fraction

$$C_a = S_a \times f$$



Average CO₂ sorption coefficient in the amorphous fraction

Henry's constant CH₄ : 0.10 to 0.14 g/(100g.Mpa)
Henry's constant CO₂ : 0.8 to 1.4 g/(100g.Mpa)

Michaels, A. S. & Bixler, H. J. Journal of Polymer Science (1961)

Naito, Y. et al. Journal of Applied Polymer Science (1996)

Von Solms, N. et al. Journal of Applied Polymer Science (2004)

Togawa, J. et al. Journal of Membrane Science (2001)

Ash, R. et al. Polymer (1970)

Flaconneche, B. et al. Oil & Gas Science and Technology - Revue d'IFP Energies nouvelles (2001)

Diffusion

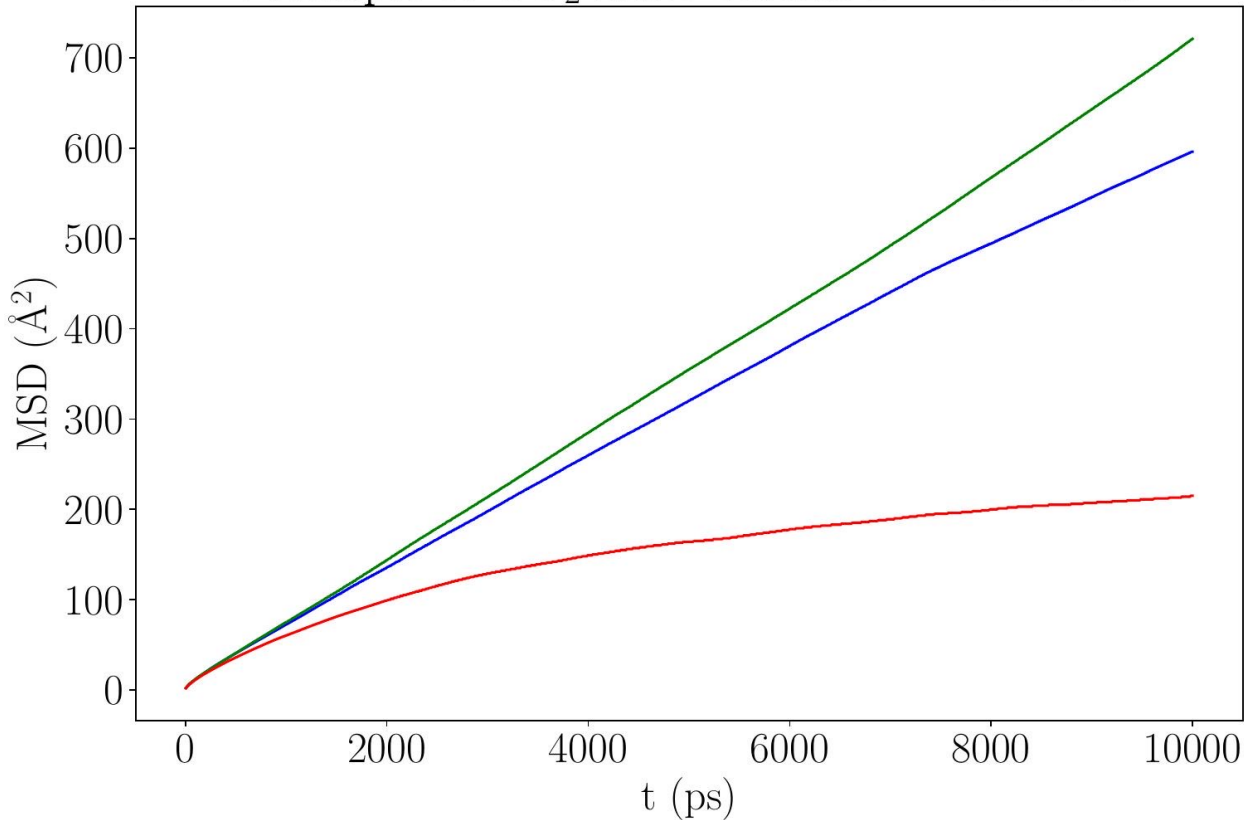
Diffusion: method

Only the x and y directions are taking into account.

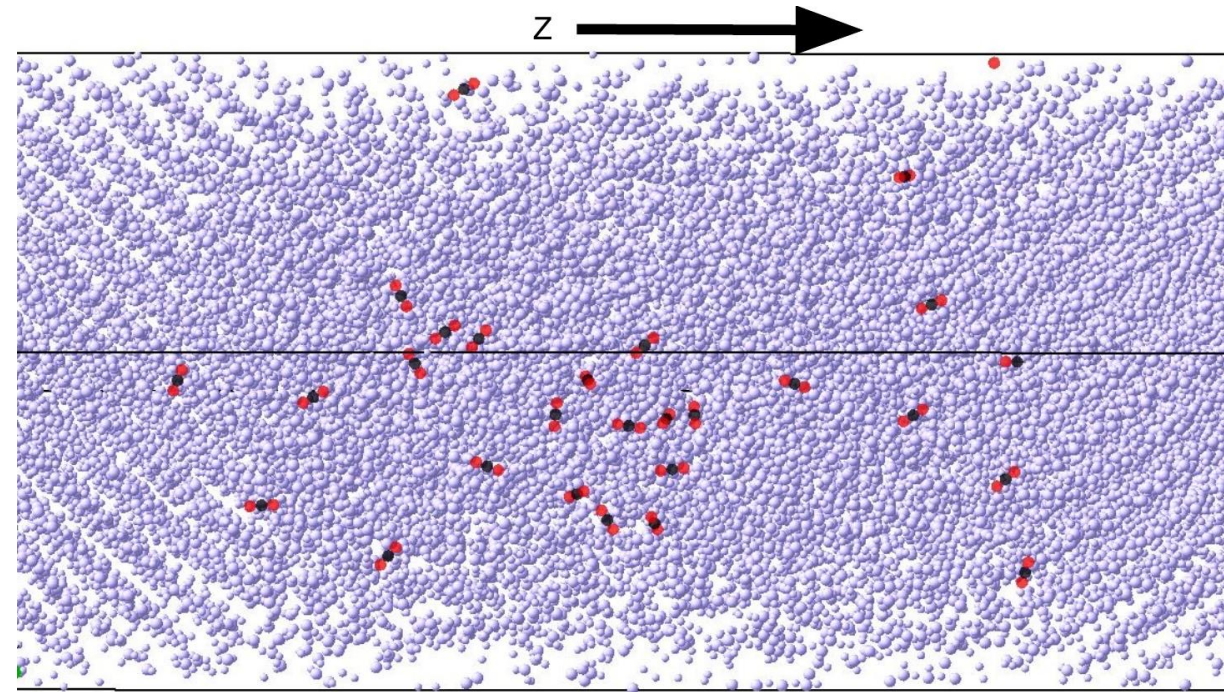
Slope of the msd and Einstein relation $D_{self} = \frac{1}{4} \lim_{t \rightarrow \infty} \frac{d}{dt} msd(t)$

$$D_{self} = D_{MS}, \quad D = QD_{MS} \text{ and } Q = \frac{\delta \ln(f)}{\delta \ln(C)}$$

Example of CO₂ MSD results $P = 40$ MPa



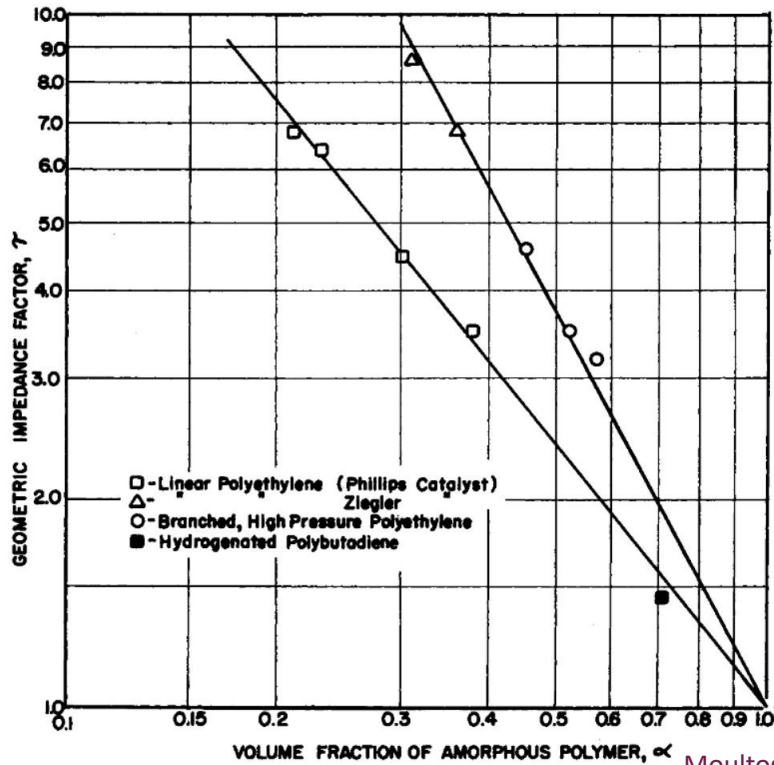
MSD, in blue and green the x and y directions, in red the z direction



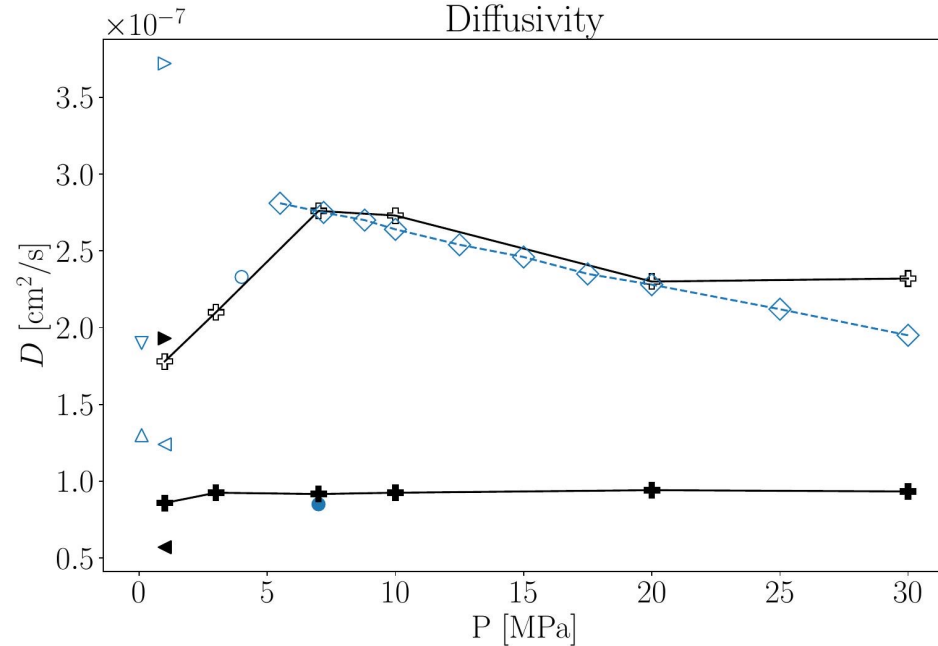
CO₂ dissolved in the amorphous region

Diffusion: result

- Michaels theory : $D = \frac{D_a}{\beta\tau}$, β is taking account by the model, so $D = \frac{D_{model}}{\tau}$.
- Michaels is giving τ with the amorphous than an equivalent amorphous polymer is natural rubber.
- τ is calibrated with Flaconnèche results, $\tau_{cal.} = 8$ (Michaels is predicting $\sim 3.5 < \tau < 6$, but TraPPE-UA overestimates D (more than 50 %), which gives $4.25 < \tau_{eff} < 9$).



τ proposed by Michaels



- +— $\bar{D}_{CH_4} \phi_a = 40\%$
- +— $\bar{D}_{CO_2} \phi_a = 40\%$
- - -◇- - - D_{CO_2} Hu $\phi_a = 40\%$
- D_{CH_4} Flaconnèche $\phi_a = 40\%$
- D_{CO_2} Flaconnèche $\phi_a = 40\%$
- △ D_{CO_2} Michaels $\phi_a = 23\%$
- ▷ D_{CO_2} Michaels $\phi_a = 57\%$
- ◀ D_{CH_4} Michaels $\phi_a = 23\%$
- ▶ D_{CH_4} Michaels $\phi_a = 57\%$
- △ D_{CO_2} Pino $\phi_a = 37\%$
- ▽ D_{CO_2} Pino $\phi_a = 44\%$

CO₂ and CH₄ diffusion coefficients with $\phi_a = 40\%$

Moulτος, O. A.; Tsimpanogiannis, I. N.; Panagiotopoulos, A. Z.; Trusler, J. P. M.; Economou, I. G. J. Phys. Chem. B 2016, 120 (50), 12890–12900

Michaels, A. S. & Bixler, H. J. Journal of Polymer Science (1961)

Flaconnèche, B. et al. Oil & Gas Science and Technology - Revue d'IFP Energies nouvelles (2001)

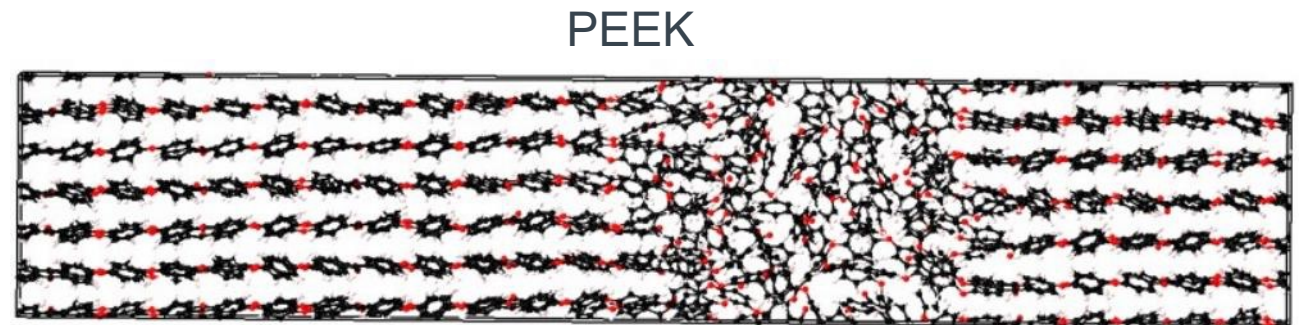
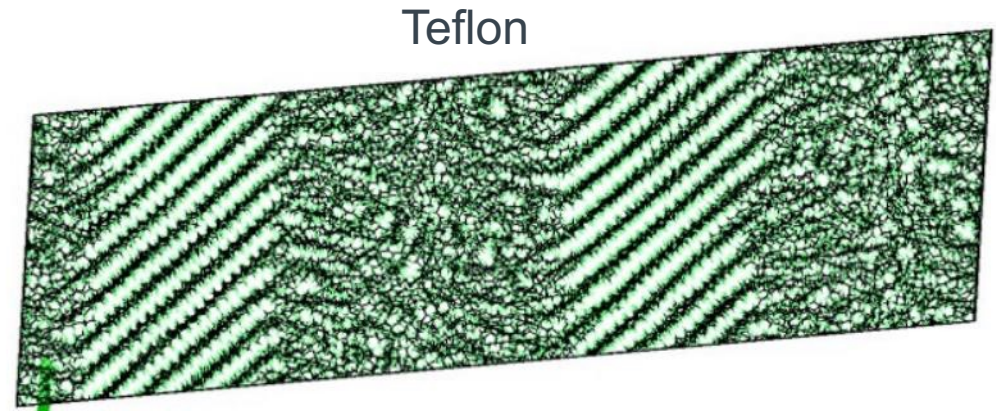
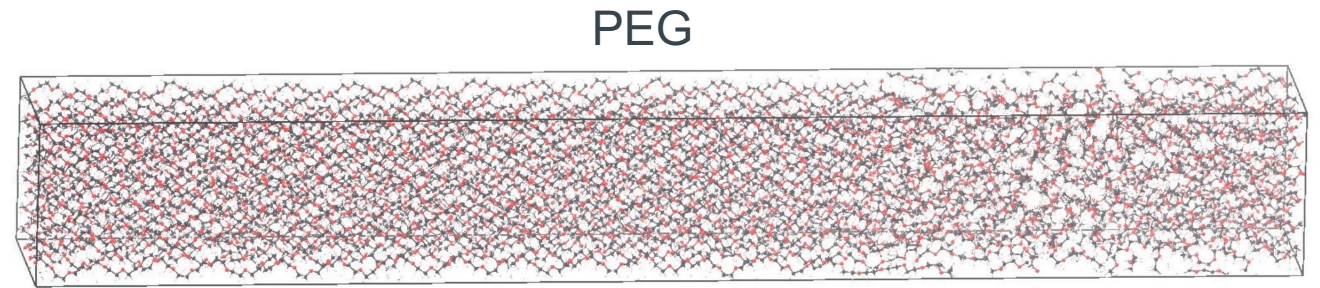
Hu, T. Permeation of High Pressure CO₂ in Semicrystalline Polymers. Phd Thesis. (Imperial College London, 2021)

Pino, M.; Duckett, R. A.; Ward, M. Polymer 2005, 46 (13), 4882–4890. <https://doi.org/10.1016/j.polymer.2005.02.118>.

Conclusion

Sorption and diffusion

- **Sorption is well predicted** with the MC/MD cycles' method.
- **Diffusivity is predicted but need one correction factor** (taking into account how the geometry of the amorphous phase in the material and maybe the forcefield). **The correction factor is species-independent.**
- **Only the crystallinity matters!** (Same for elastic properties in previous webinar).



Highlighted *MedeA* Modules

MedeA Environment: with high productivity integrated tools, property prediction using application-specific property modules in conjunction with state-of-the-art simulation engines, reproducible workflows as graphical flowcharts for multi-stage protocols

MedeA JobServer & TaskServer: central unit to manage, monitor, and archive your calculations efficiently and consistently. highly flexible workflow maximizes productivity, on systems with heterogeneous hardware, and different operating systems.

MedeA Amorphous Materials Builder: lets you efficiently create condensed phase models based on system chemical composition and target density. It eliminates the need for lengthy mixing and amorphization simulations through realistic sampling of the translational, rotational, and conformational degrees of freedom of component species.

MedeA Polymer builder: Creates models of isolated polymer chains, providing a foundation for building more complex models. Examples include bulk polymers, blends, solutions, or multiphase systems incorporating one or more interfacial regions.

MedeA Thermoset Builder: Applies state-of-the-art methods for creating complex topologies of polymer networks in order to create strain-free molecular models with experimentally observed crosslink densities.

MedeA LAMMPS: Provides flexible calculation setup and analysis capabilities to unlock the power of LAMMPS.

MedeA Gibbs: The reference approach for fluid properties and sorption.

Question and Answer Session



Dr. Marianna Yiannourakou

Materials Design



Dr. Boris Belin

Materials Design



Dr. Dave Rigby

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Questions about Materials Design Webinars

Katherine Hollingsworth

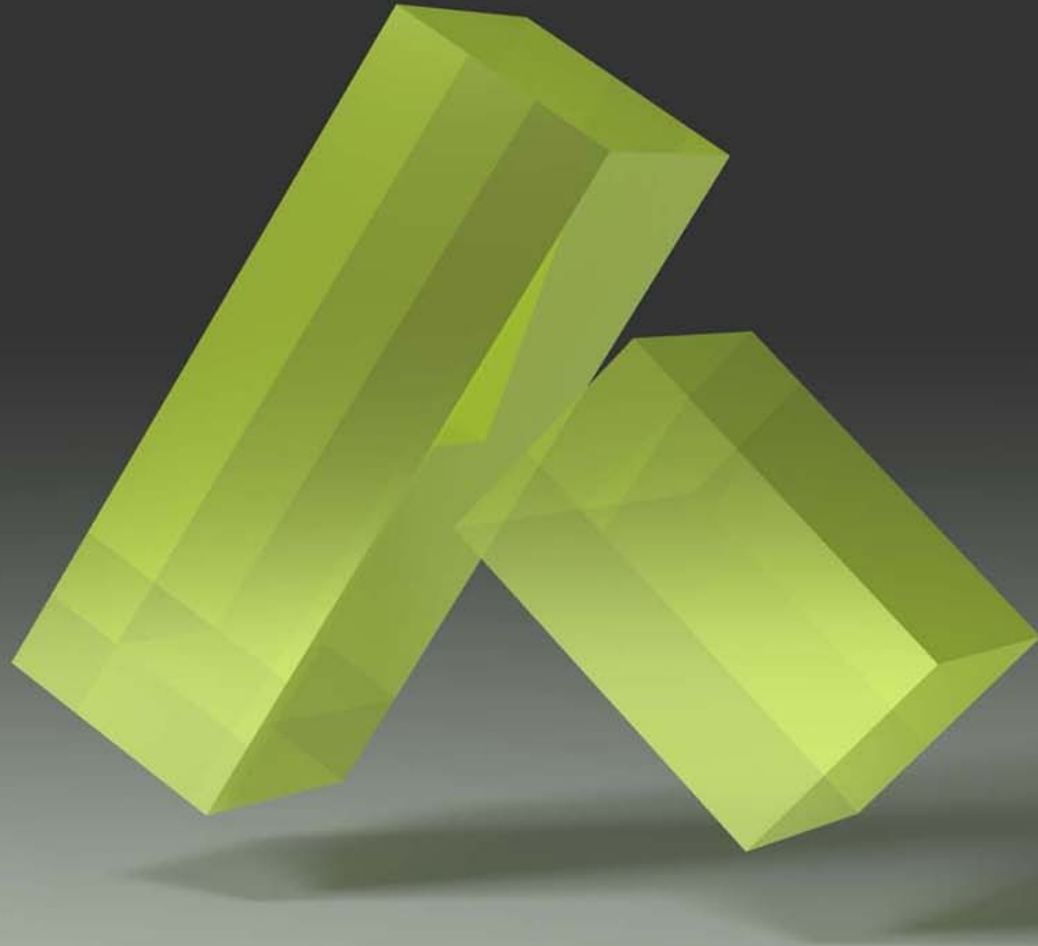
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