



materials design


Li-Ion Batteries and Beyond: Driving Next-Generation Energy Storage with *MedeA*[®]

Taylor Juran
Materials Design

Jan 2021



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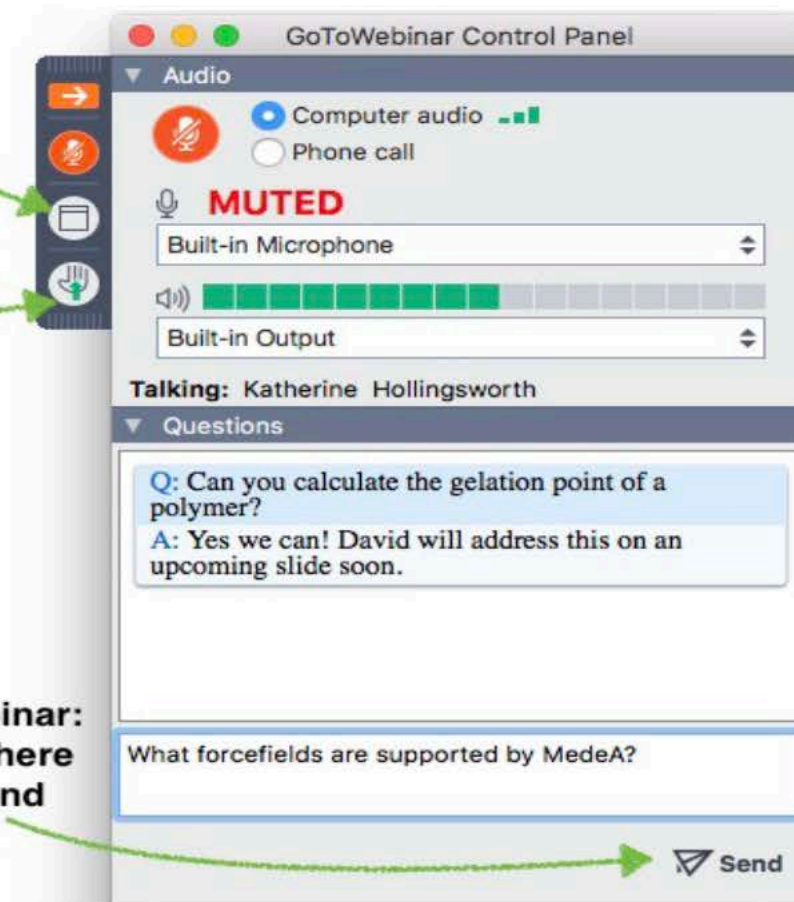
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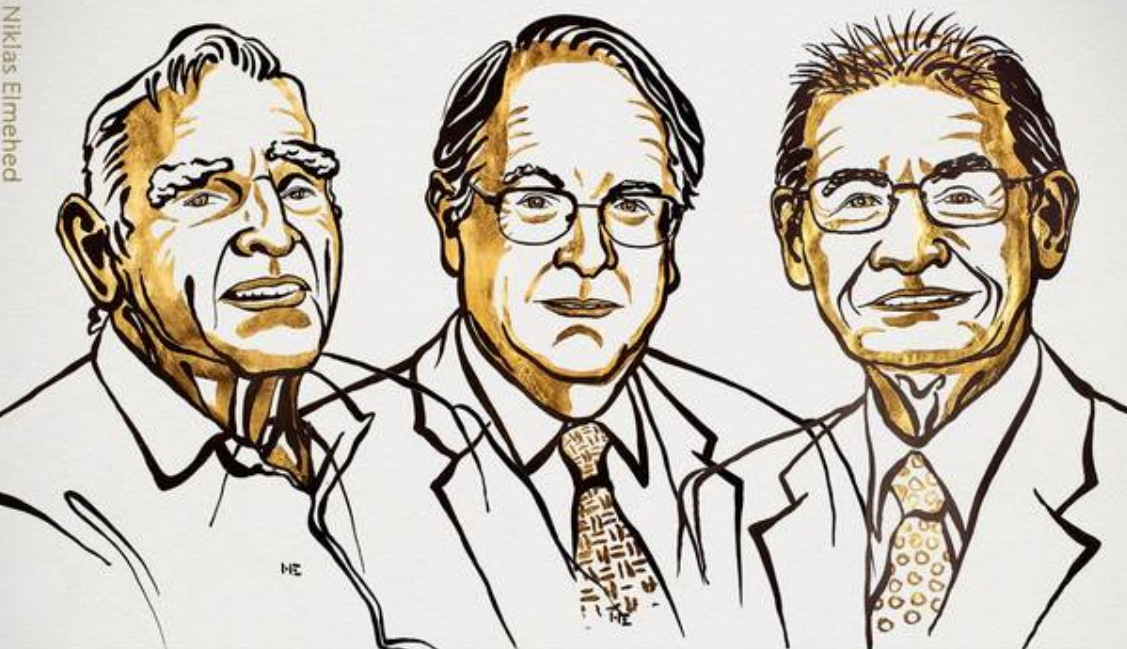
Outline

- Batteries Today
- Cluster Expansion of Li_xNiO_2
- Li_xNiO_2 Voltage Profile
- Multivalent Ion Batteries
- Intercalation Site Identification
- Electronic Properties
- Diffusion Barriers

Battery conversations become mainstream

Illustrations: Niklas Elmehed

THE NOBEL PRIZE
IN CHEMISTRY 2019



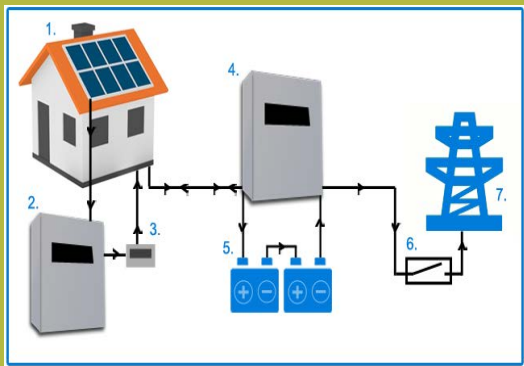
John B. Goodenough M. Stanley Whittingham Akira Yoshino

“for the development of lithium-ion batteries”

THE ROYAL SWEDISH ACADEMY OF SCIENCES



Branches of batteries



Vanadium-Redox Flow

Zinc-Polyiodide Flow

Organic Aqueous Flow

Na-Metal Halide

Na-ion

Mg-ion

Mg-Li Hybrid

Zinc-Manganese Oxide



Li-ion

Li-S

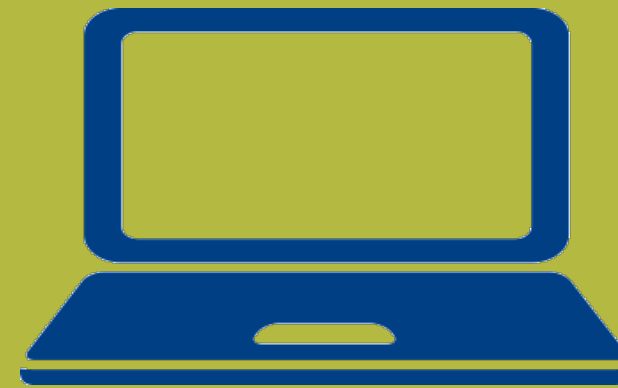
Li-Air

Li-Metal

Mg-ion

Ca-ion

Toyota^[1-3]



Li-ion

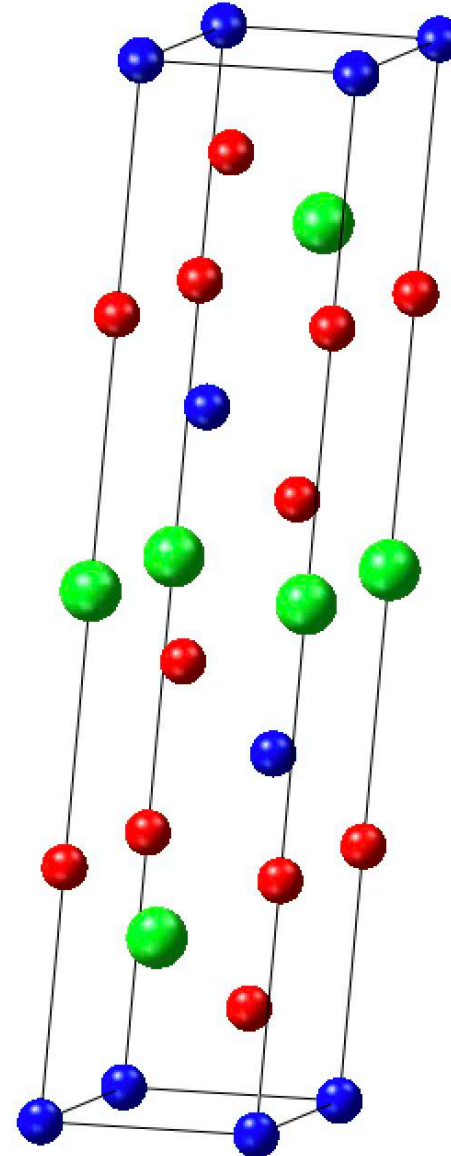
[1] Joule, Volume 3, Issue 3, P782-793, MARCH 20, 2019

[2] Nat Energy 5, 1043–1050 (2020).

[3] Front. Chem., 20 February 2019

'Li is the new oil'

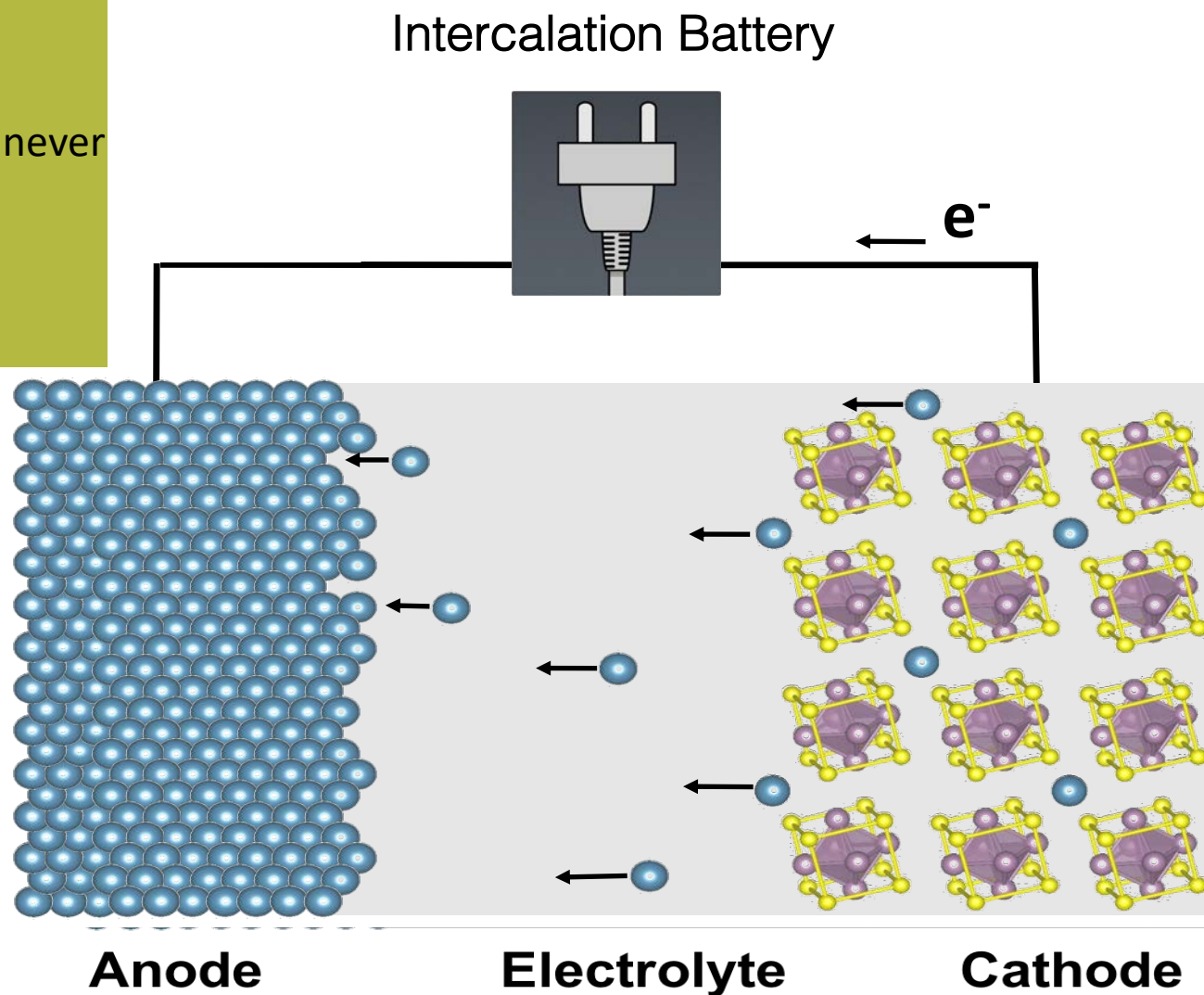
- Li is the lightest metal
- Most electropositive metal
- Energy density – 260 Wh/kg
- High cyclability – over 500 cycles
- Limited by the cathode
- LiCoO_2



We need new battery chemistries

“If you want something you've never had
You must be willing to do something you've never
done.”

— Thomas Jefferson.





MedeA, take the wheel

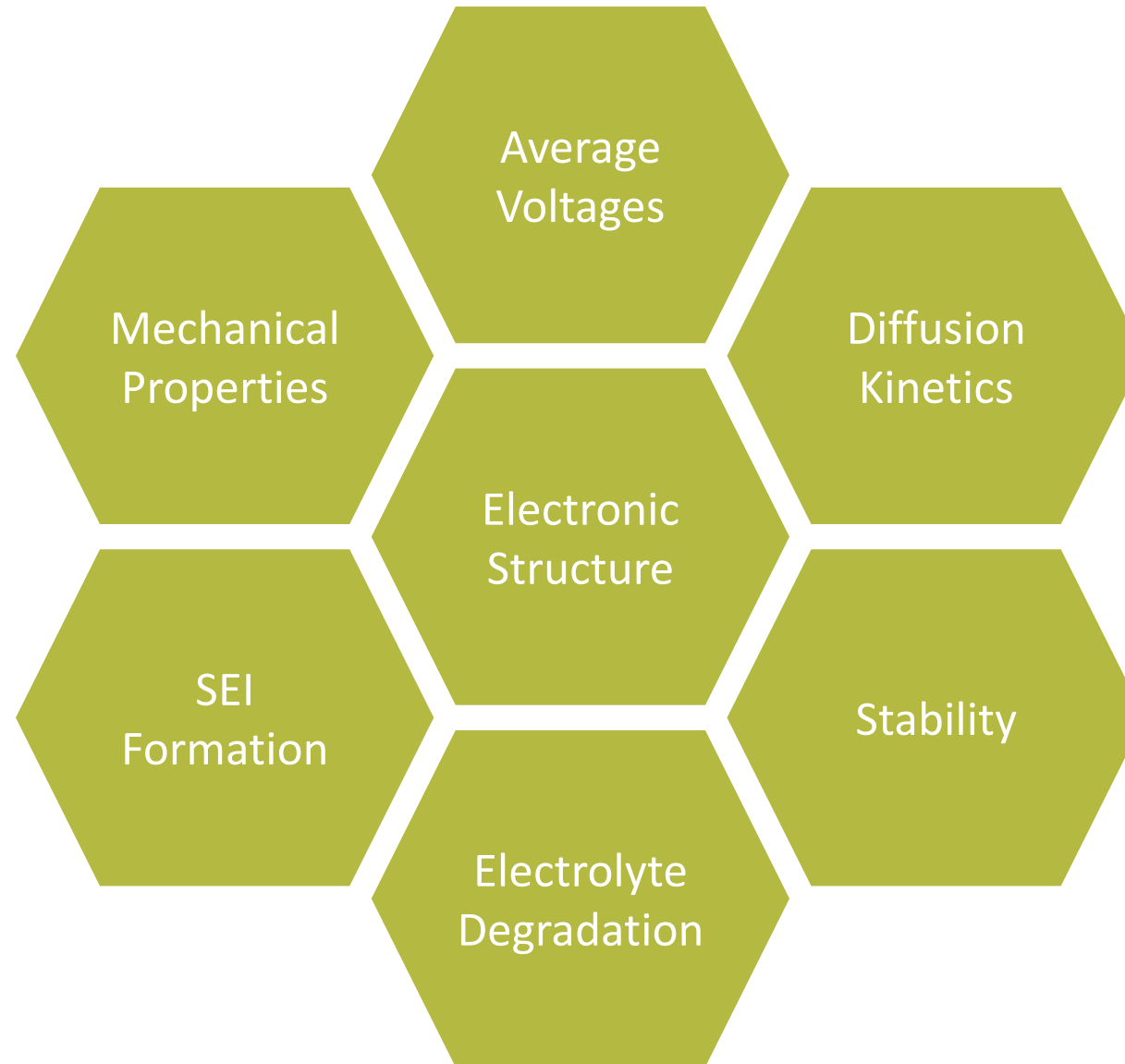
Drive the science forward, instead of spending valuable research time writing bad code

Modeling becomes more accessible

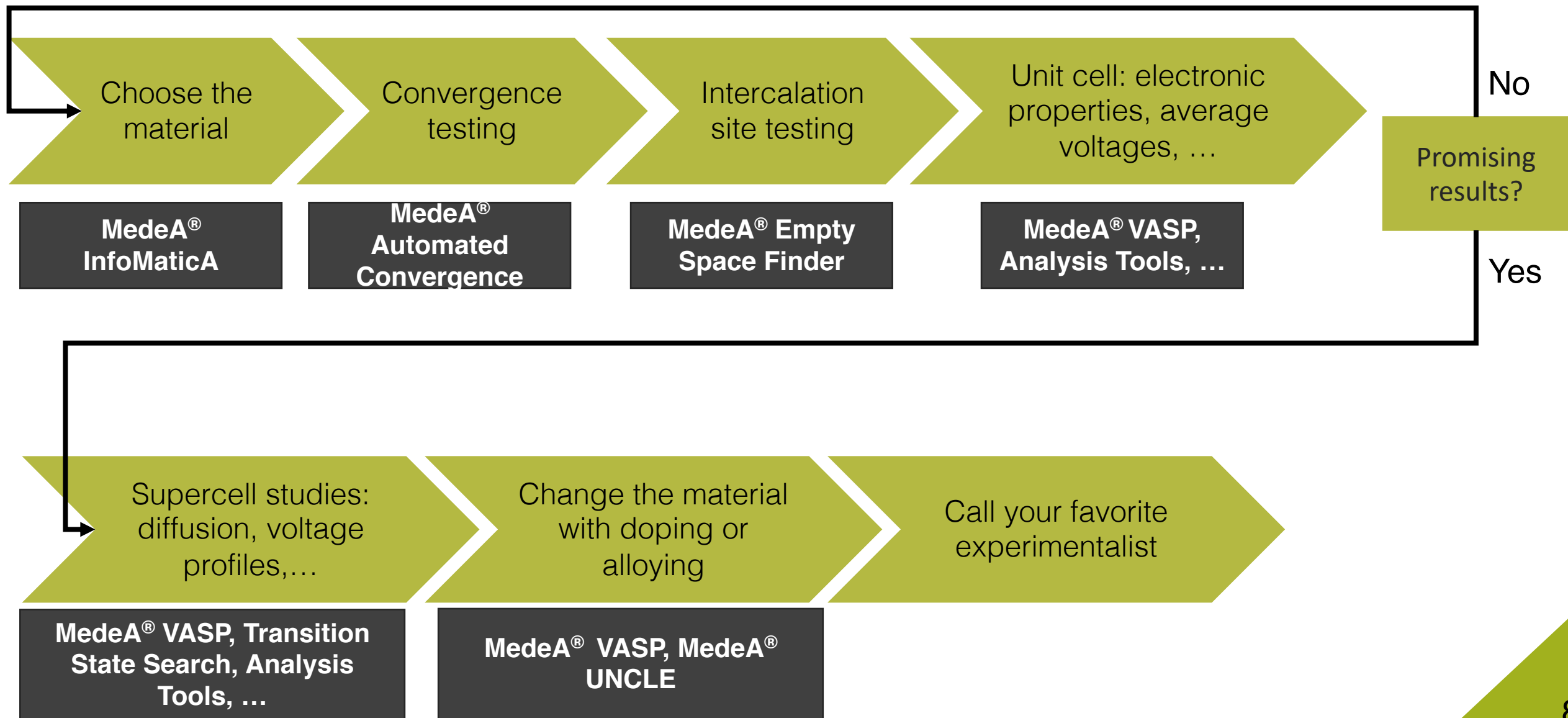
Study a broader range of batteries

Reserve experimental efforts for only the most promising battery candidates

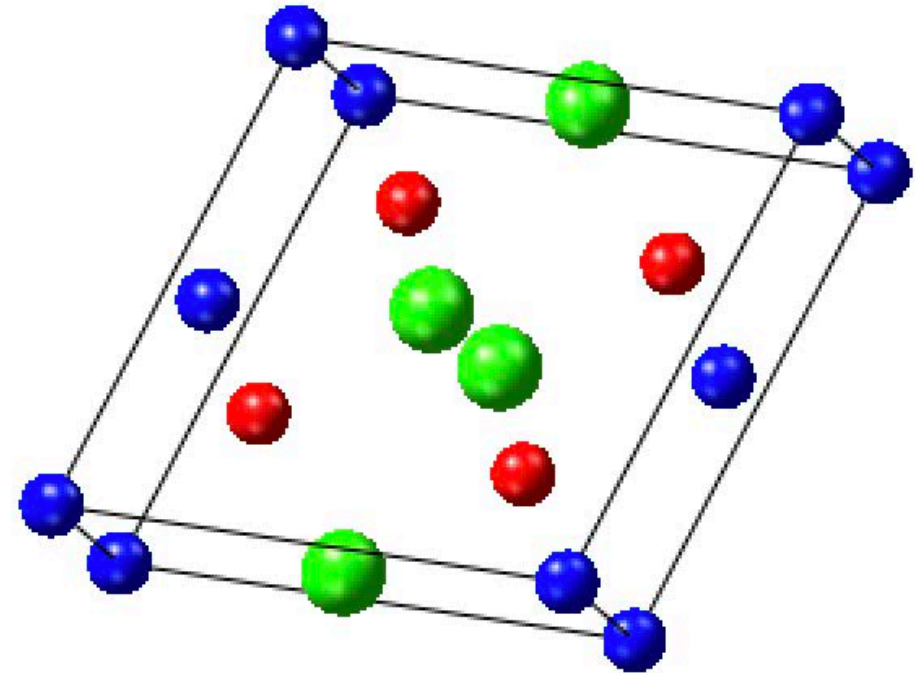
Applying *MedeA* to battery material research



A computational study of anode/cathode pairs: The outline



- LiCoO_2 is a commonly commercialized battery material - since the 1980's
- Mixing the transition metal M in LiMO_2 allows battery properties to be tuned
- Ni provides a favorable capacity
- Electrochemical properties are heavily dependent on cathode structural changes during cycling^[1]



LiNiO₂ structure: Li (blue), Ni (green), O (red)

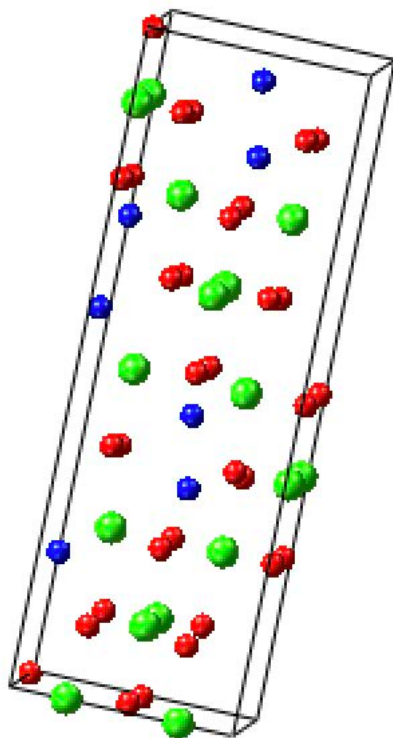
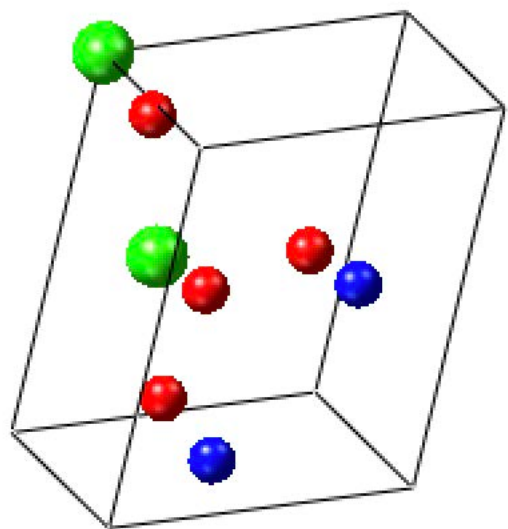
Li_xNiO_2 different phases, and ranges they are stable

Rock Salt
Fm-3m
 $0 \leq x \leq 0.5$

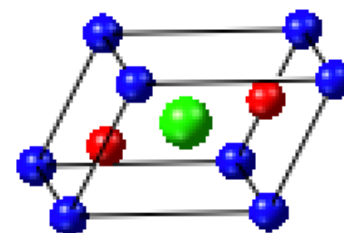
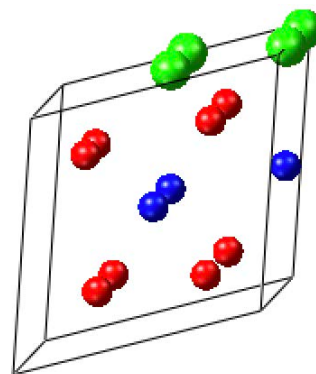
Spinel
Fd-3m
 $0.5 \leq x \leq 0.6$

Layered
Hexagonal
R-3m
 $0.4 \leq x \leq 1$

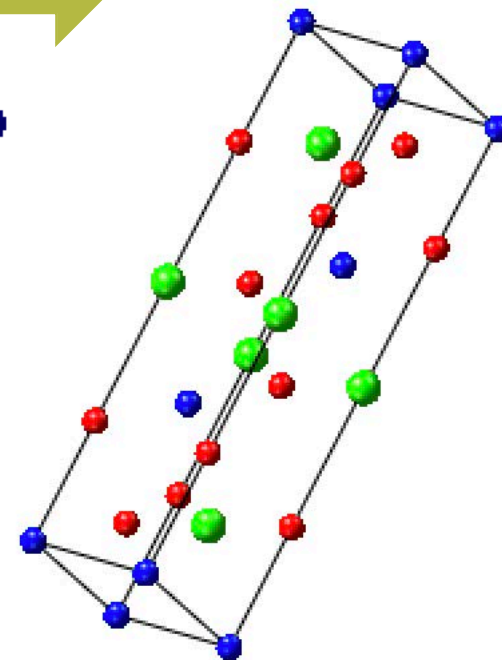
Increasing Li



Layered
Monoclinic
C2/m
 $0.39 \leq x \leq 0.69$

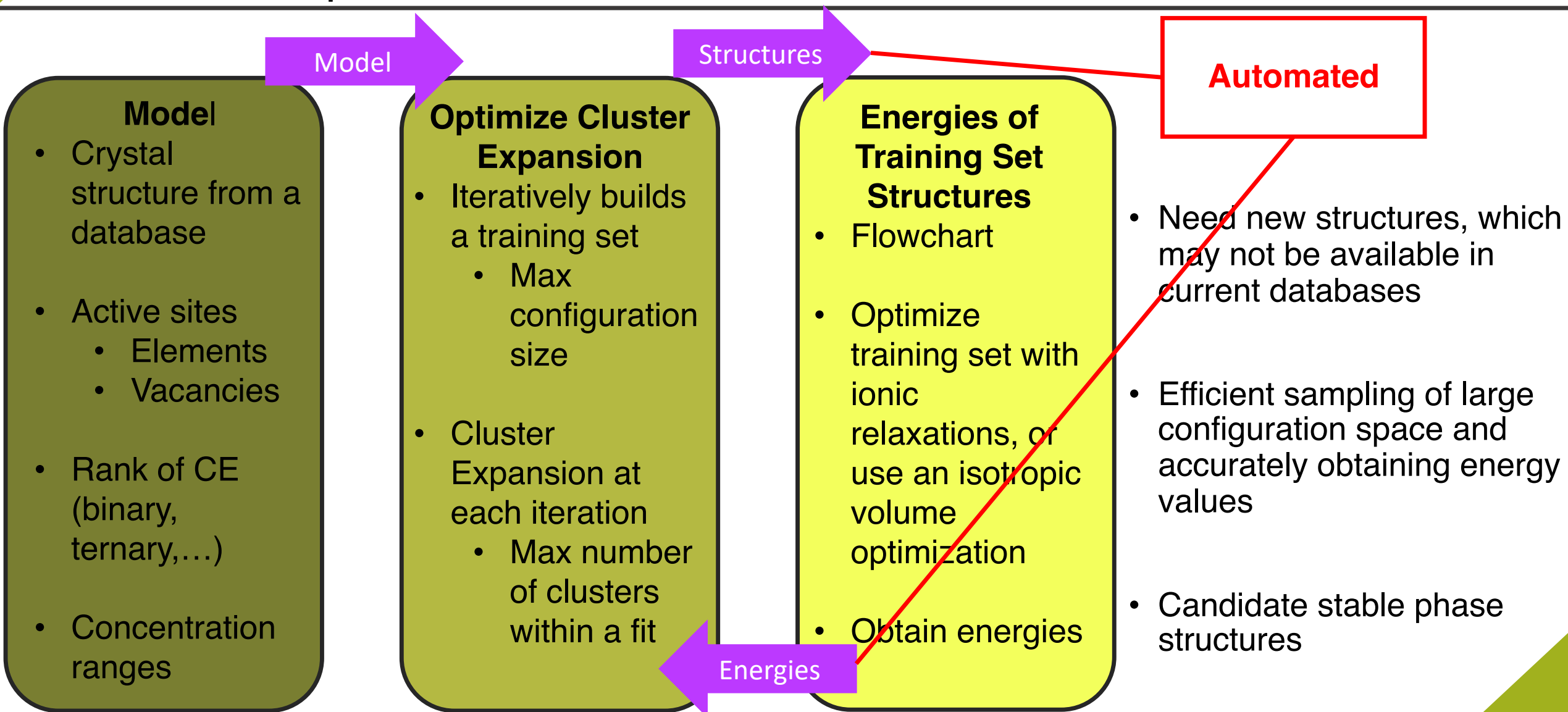


Hexagonal primitive cell
to raised symmetry R-3m

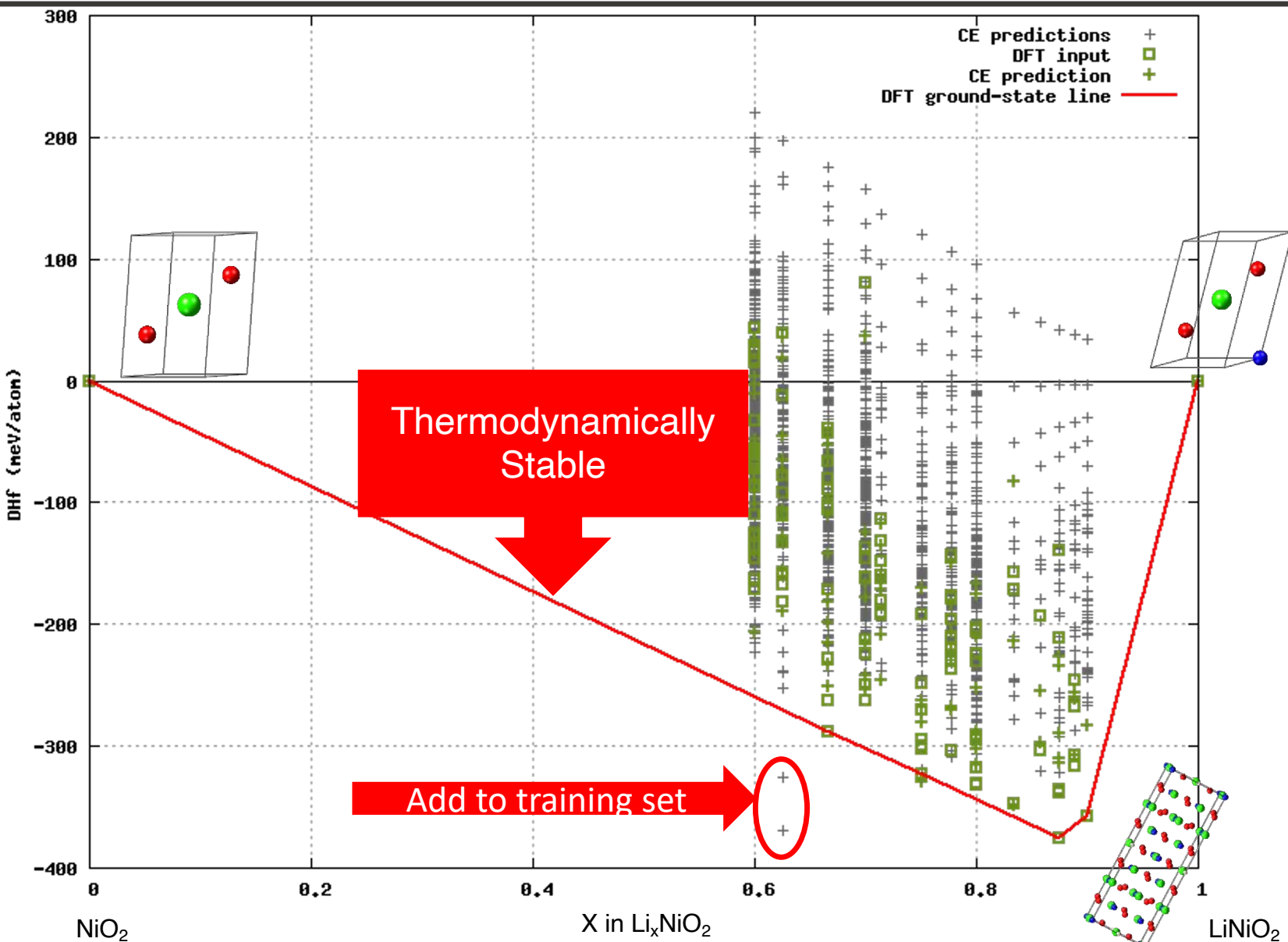


LiNiO_2 structure: Li (blue), Ni (green), O (red)

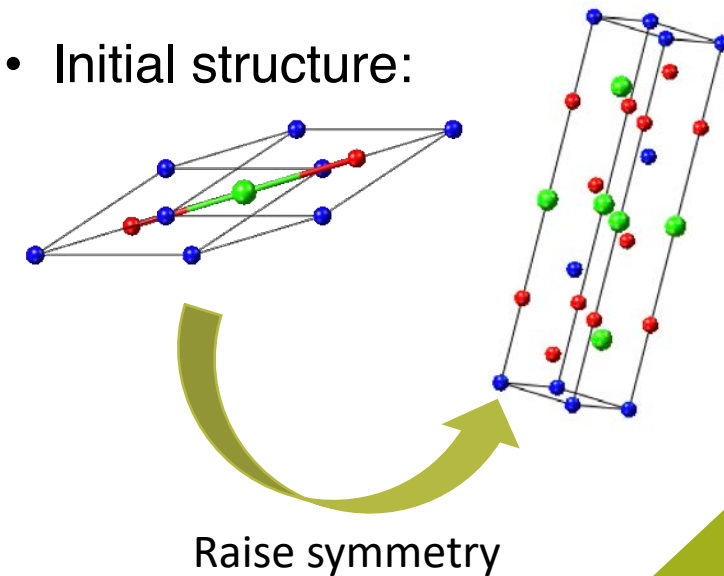
MedeA UNCLE – Identifying new cathode structures with Cluster Expansion



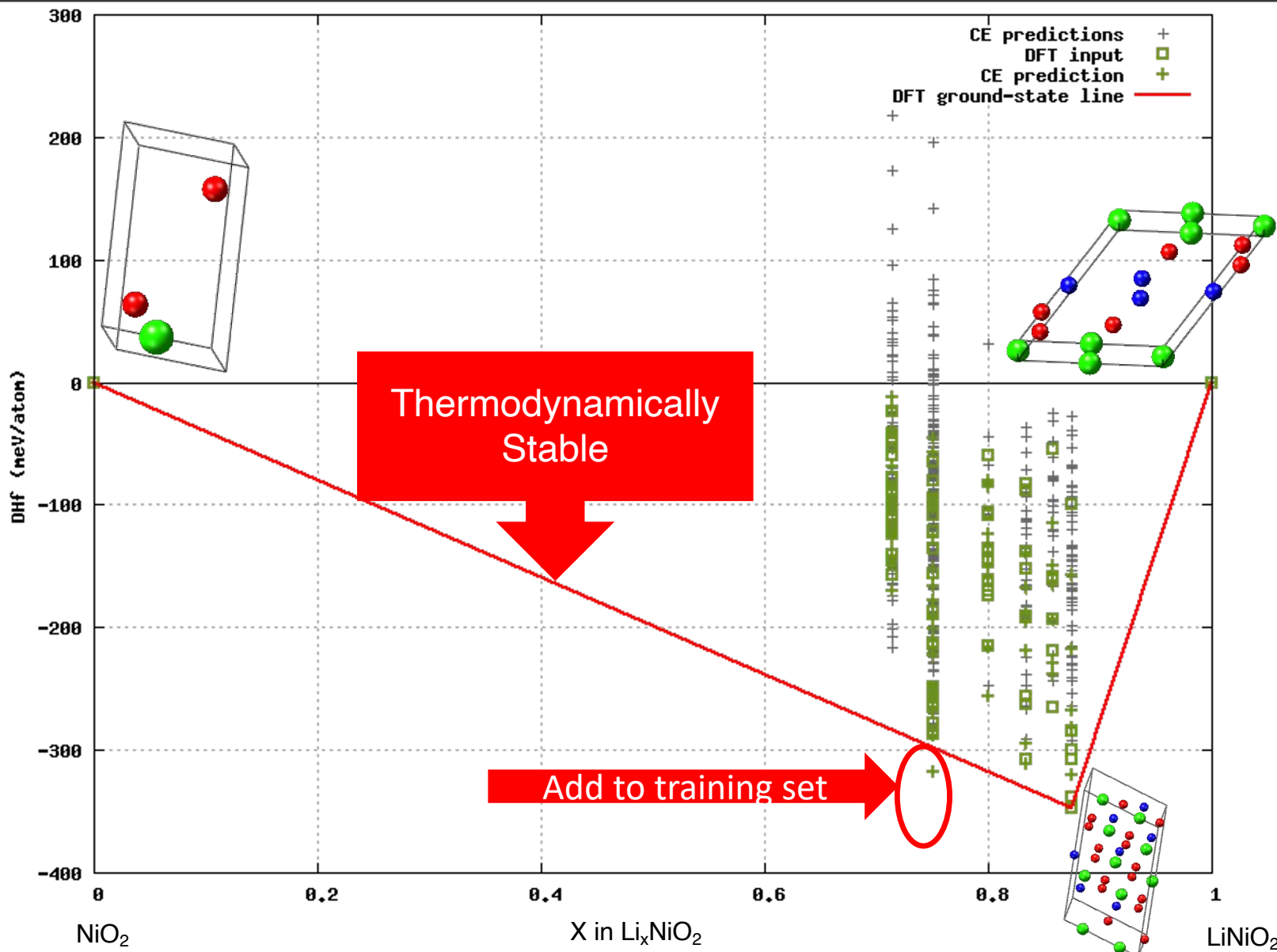
Li_xNiO_2 Layered Hexagonal (R-3m)



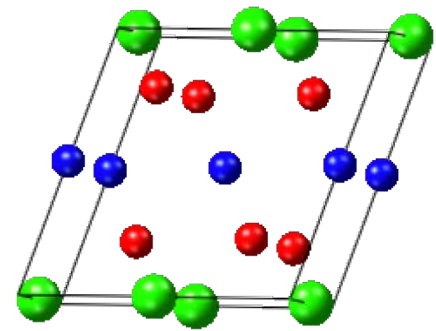
- Max generations = 200
- Max clusters in fit = 20
- Cross Validation Score (CVS) ≤ 7.7 meV/site
- Initial structure:



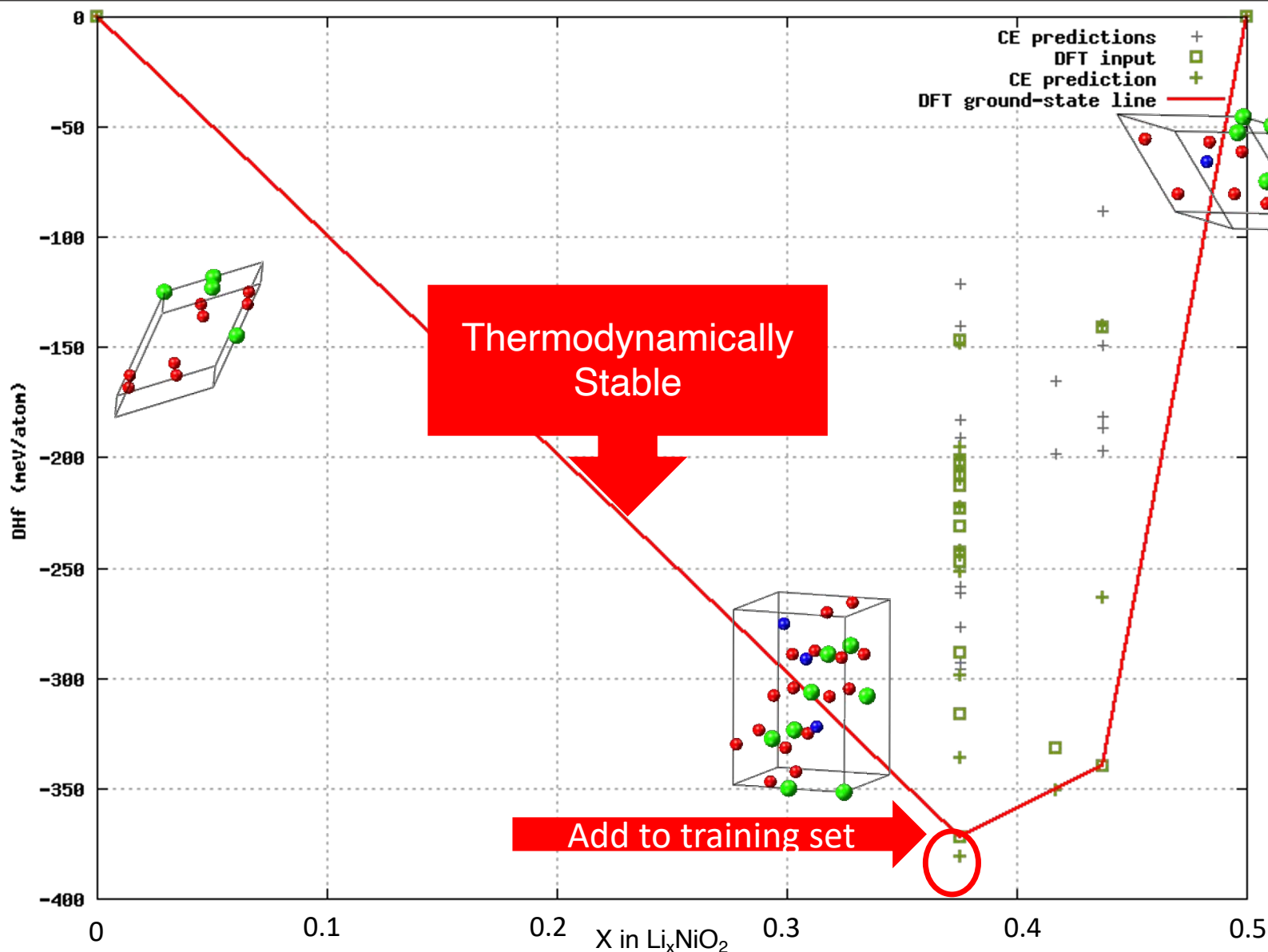
Li_xNiO_2 Layered Monoclinic (C2/m)



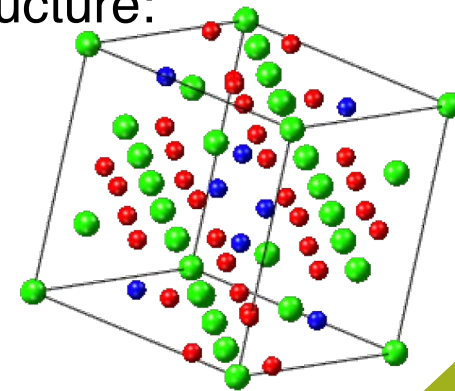
- Max generations = 200
- Max clusters in fit = 20
- Cross Validation Score (CVS) ≤ 35 meV/site
- Initial structure:



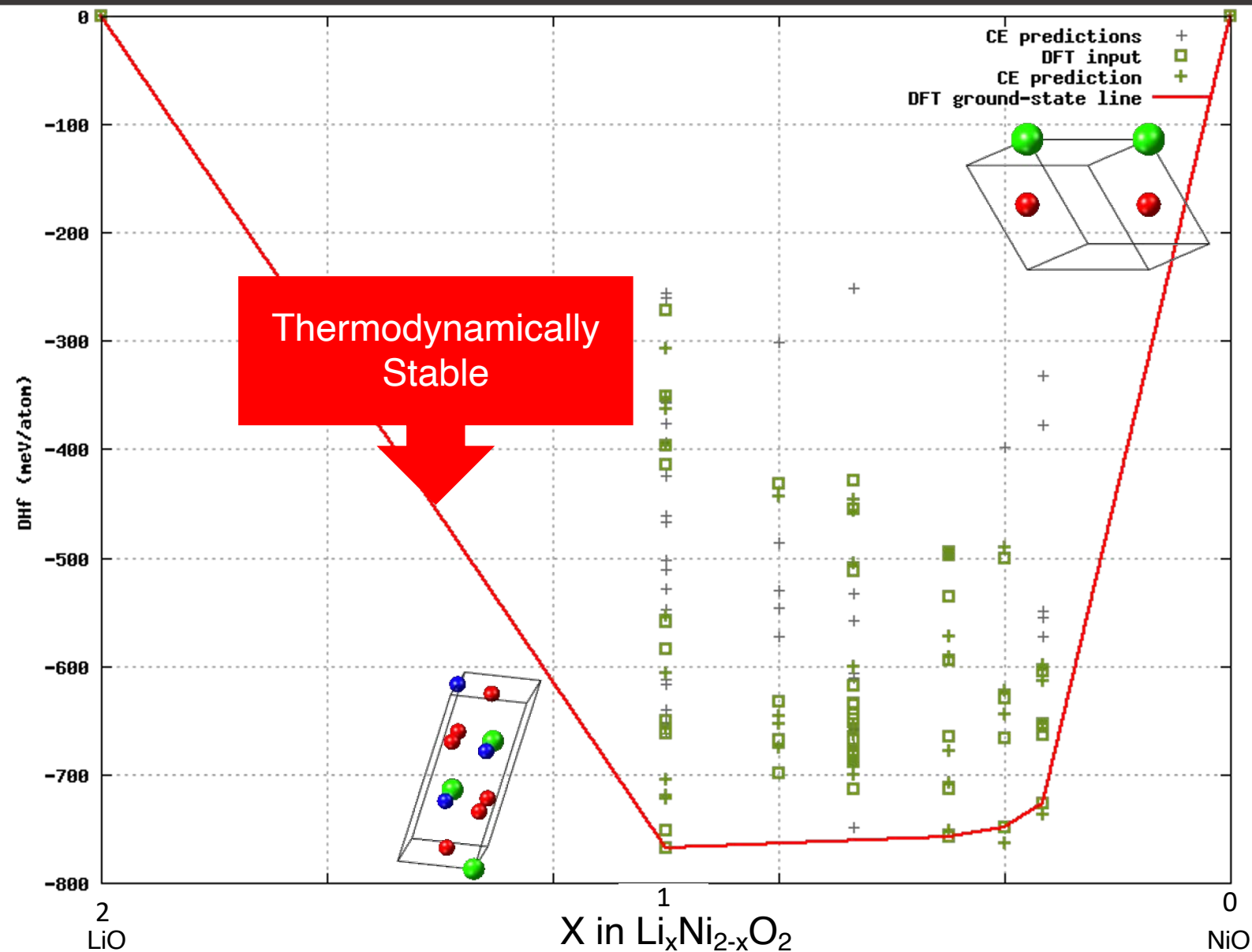
Li_xNiO_2 Spinel (Fd-3m)



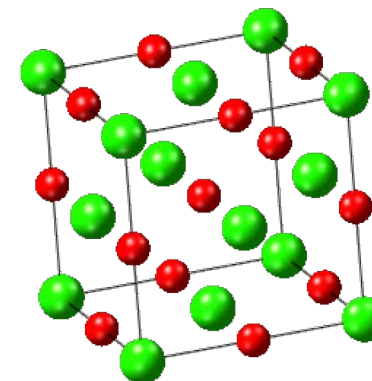
- Max generations = 200
- Max clusters in fit = 20
- Cross Validation Score (CVS) ≤ 22 meV/site
- Initial structure:



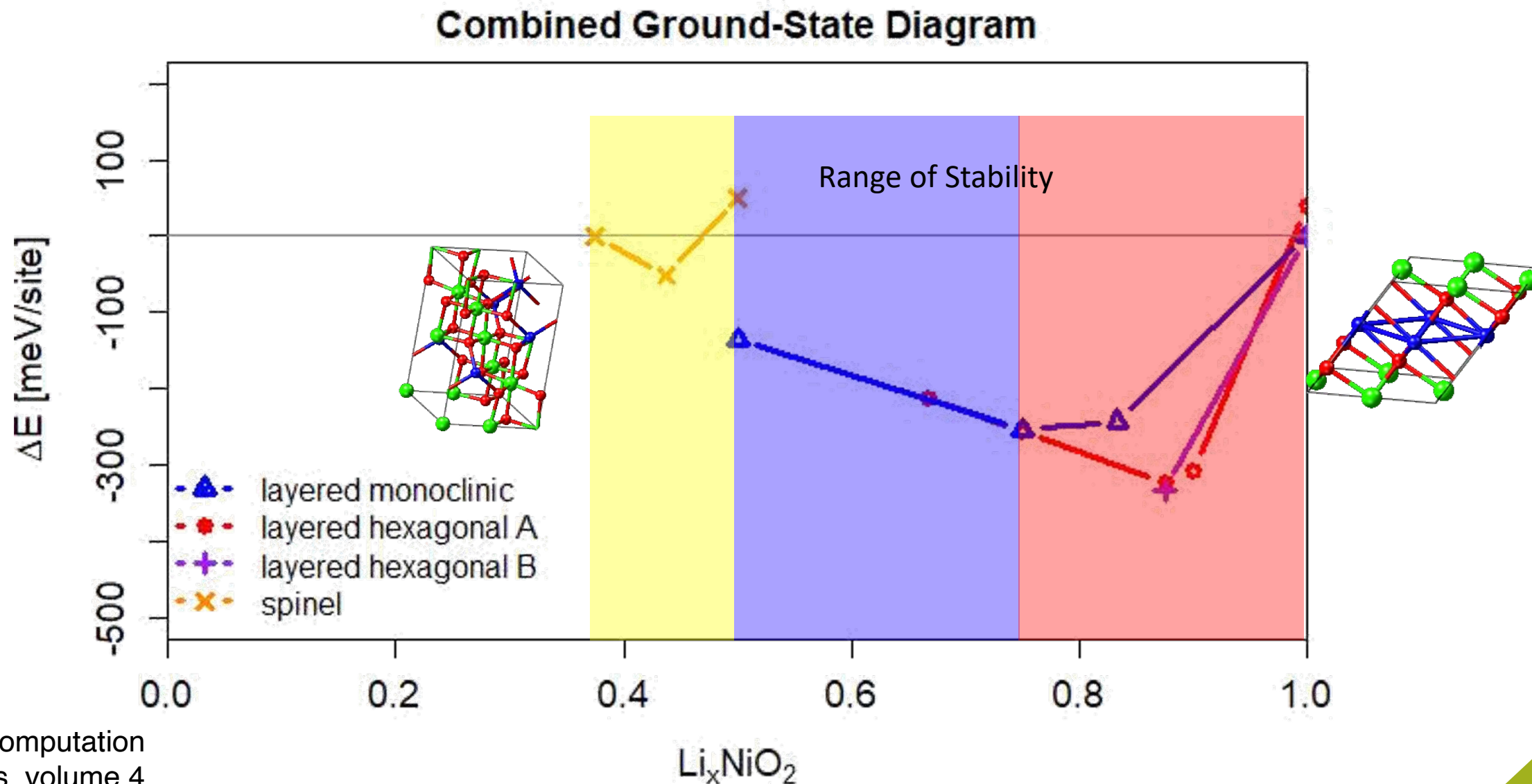
2 Phase Regime: X in $\text{Li}_x\text{Ni}_{2-x}\text{O}_2$ Rock Salt (Fm-3m)



- Max generations = 200
- Max clusters in fit = 20
- Cross Validation Score (CVS) ≤ 26 meV/site
- Initial structure:

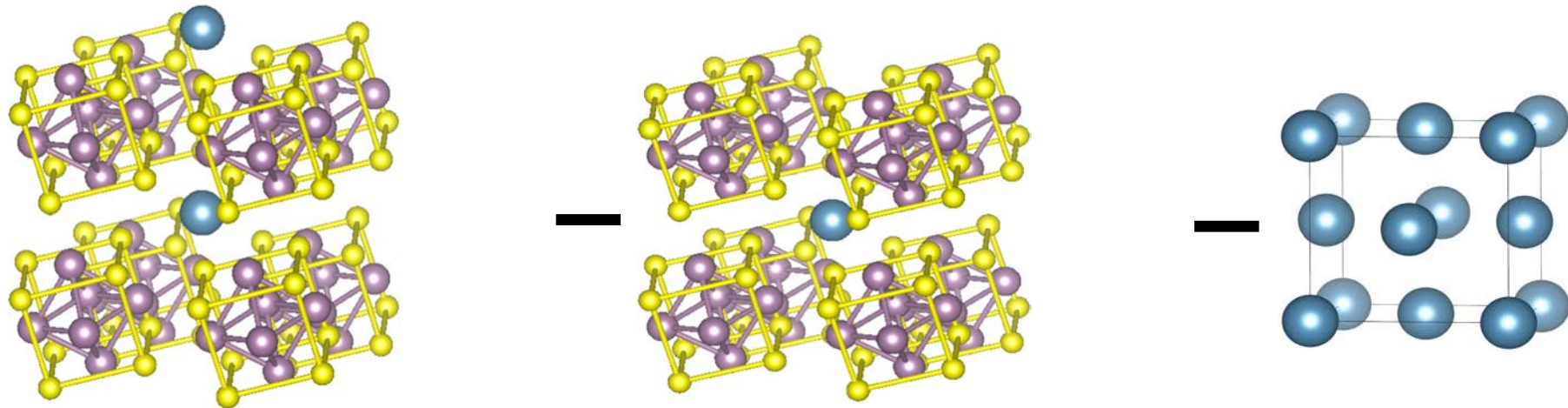


Li_xNiO_2 Combined Ground-State Diagram



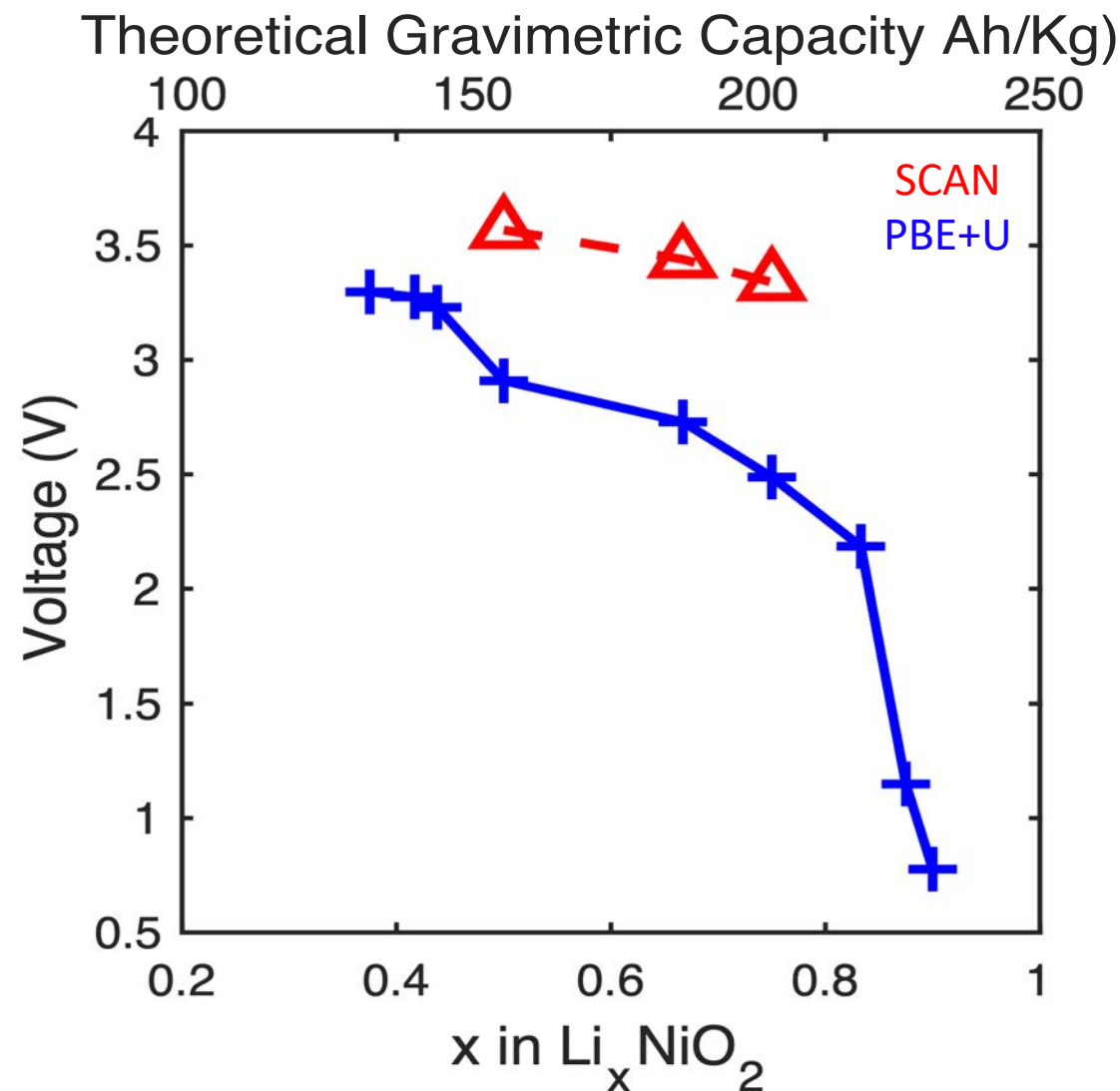
Voltage calculations

$$V = - \left(\frac{E(M_{x_2} \text{ Cathode}) - E(M_{x_1} \text{ Cathode}) - E((x_2 - x_1) \mu^{\text{H}})}{N_{\text{electrons}}} \right)$$



Voltages calculated for the previously identified stable structures

- PBE +U, U = 7.2
- Voltage is underestimated compared to experiment
- Voltages match those computed with PBE in a previous study
- SCAN should increase the voltage, and possibly lead to a better experimental match
- Compares well with other studies



Functionals do matter, but we must work within our means



- Toyota and Honda: LDA and GGA
 - Local Density Approximation
 - Perdew-Burke-Ernzerhof (PBE)
 - PBE + Hubbard-U (PBE+U)
- BMW and Mercedes-Benz: Meta-GGAs
 - Strongly Constrained and Appropriately Normed semi-local density (SCAN)
 - Modified Becke-Johnson (mBJ)
- Rolls Royce – Hybrids
 - Hyde-Scuseria-Ernzerhof (HSE)

Li is 'the new oil', but Ca might fill in the gap

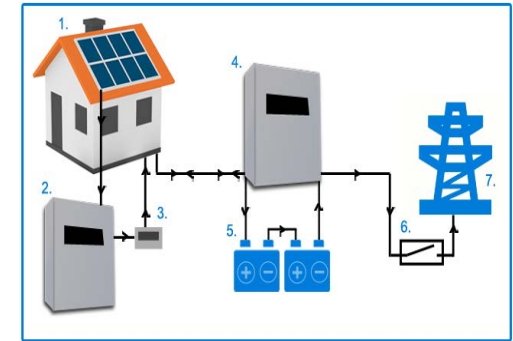
- Range of battery chemistries to select from
- Better batteries:
 - Energy density – Capacity and voltage
 - Charge/Discharge Rate – mobility of charge carriers
 - Lifetime – stability, phase transitions, swelling and cracking



An alternative – when portability is not a concern

- Multivalent Ion Batteries

- Reduced cost
- Low environmental impact
- Improved safety

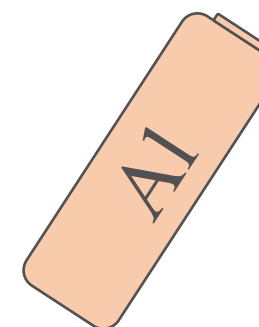
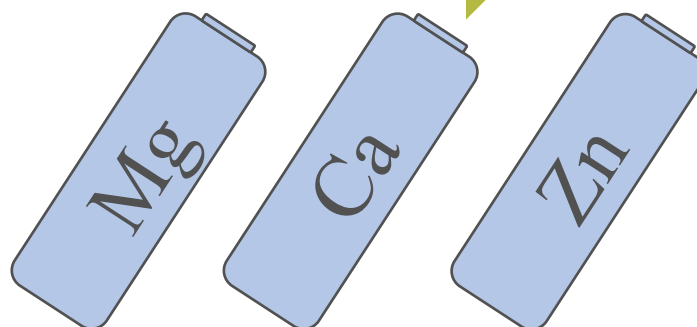
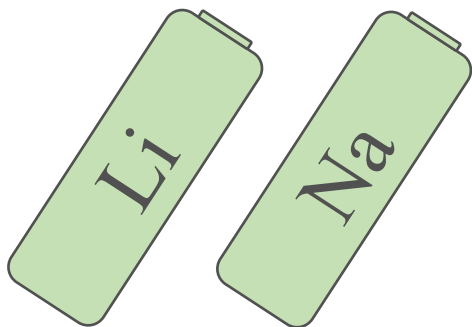


Are two better than one?

Increased Valency

Increased Capacity

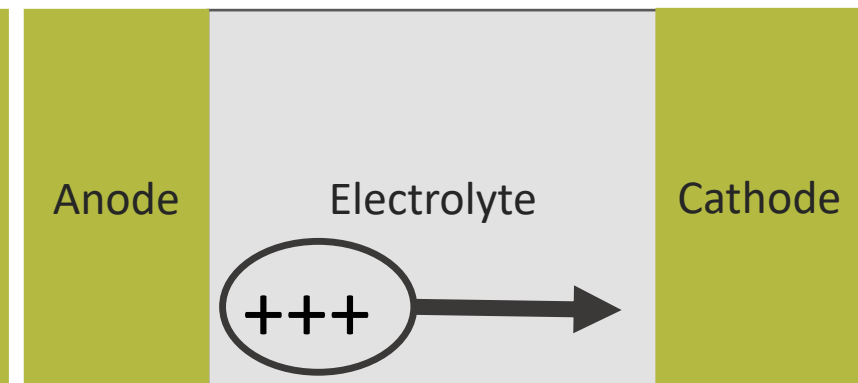
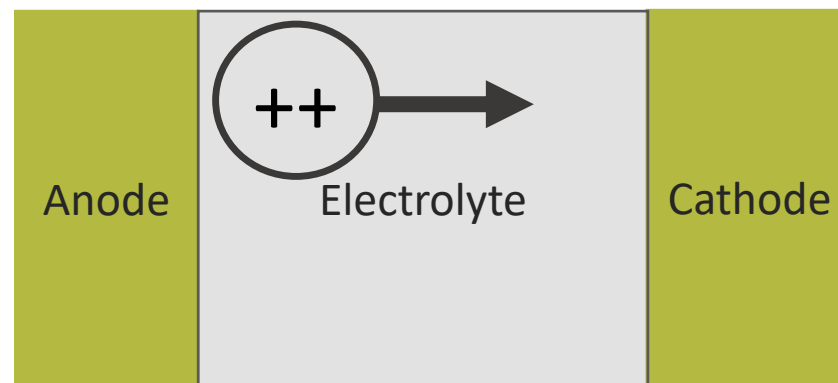
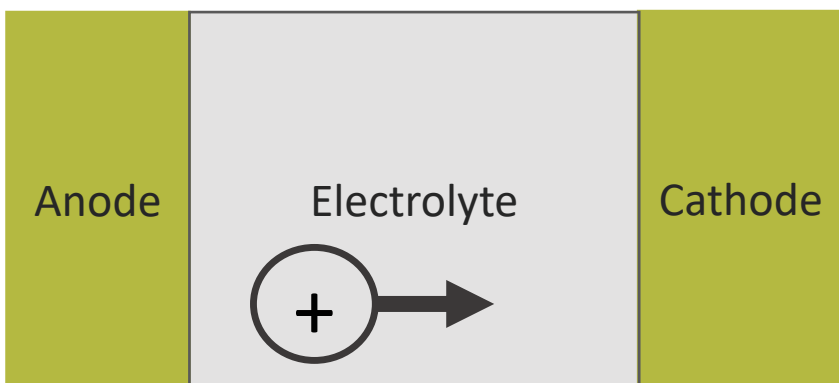
Increased Energy Density



Monovalent

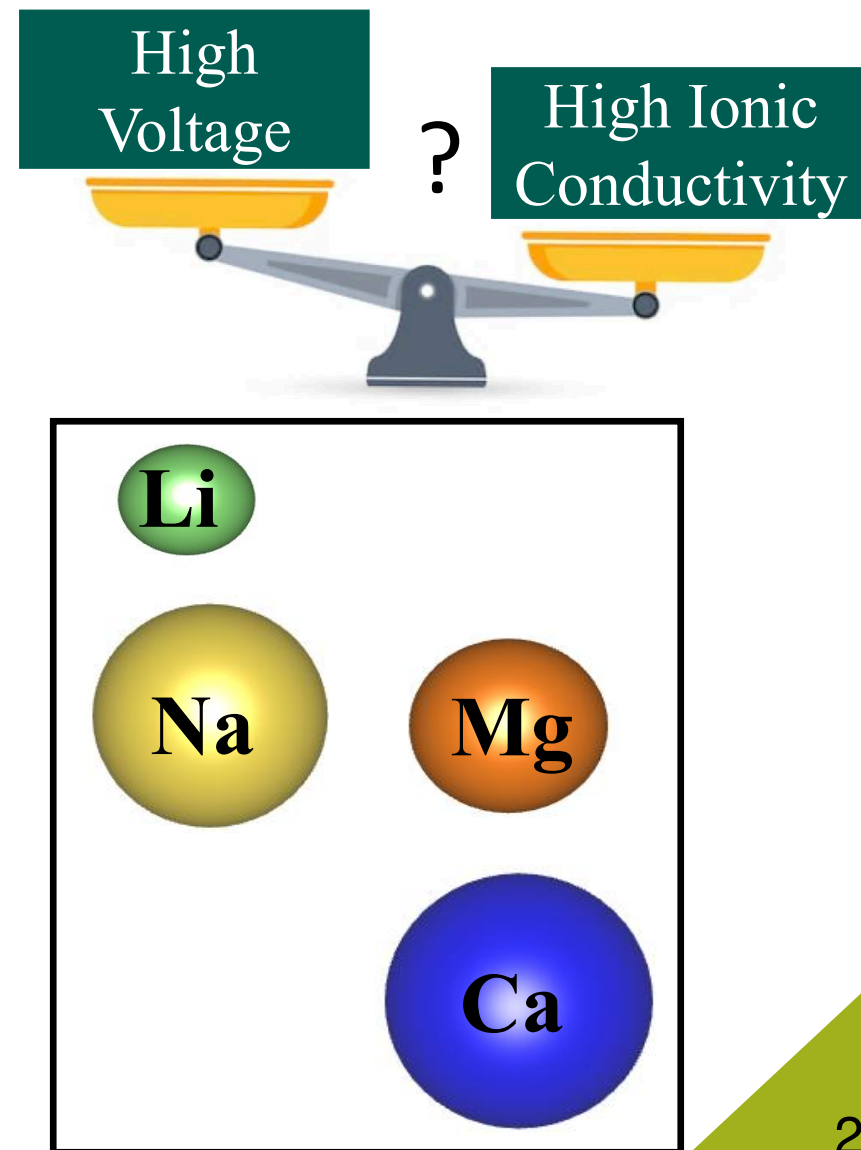
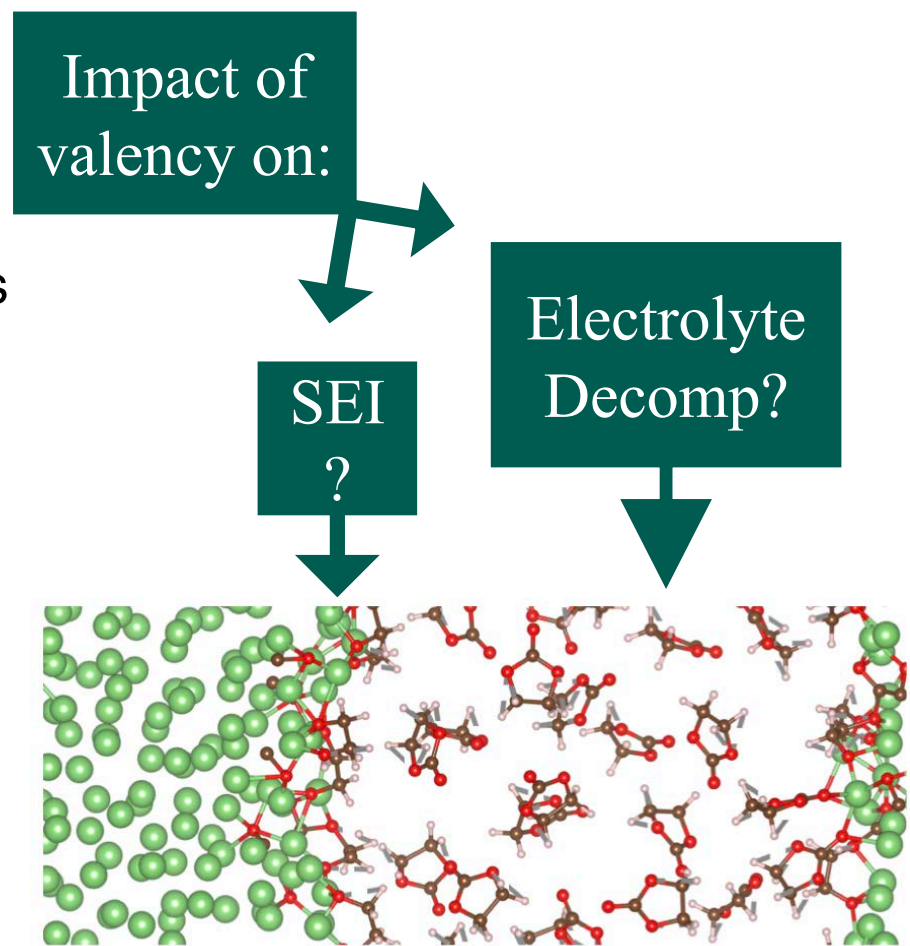
Divalent

Trivalent



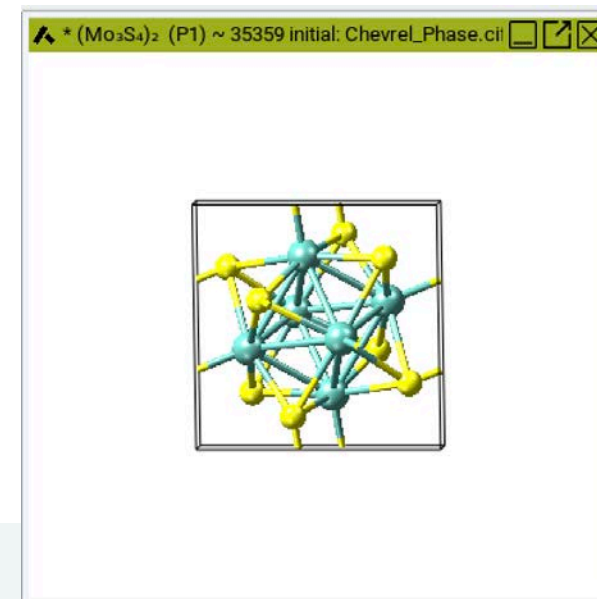
Major drawbacks of multivalent ion batteries

- Primarily limited by the cathode
- Desired cathode qualities:
 - High energy density
 - Cyclability
 - Good diffusion kinetics
- Lack of information regarding the impact of valency on electrolyte and SEI



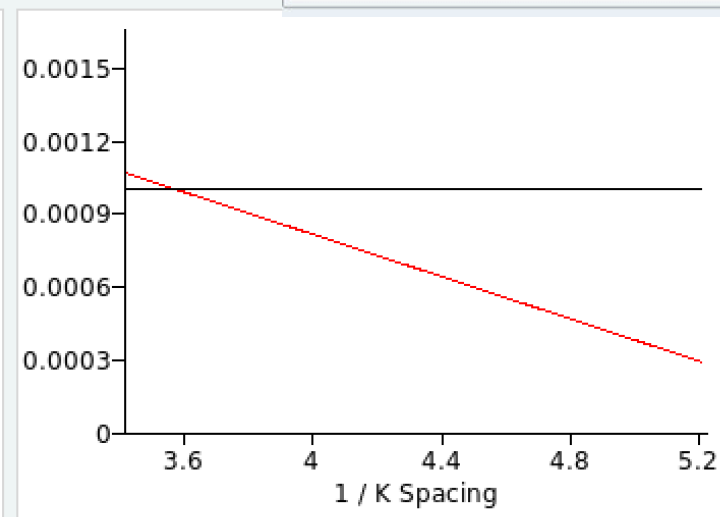
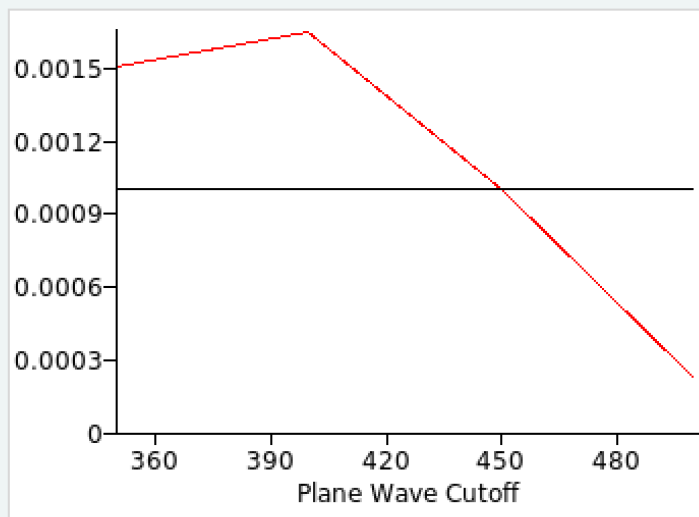
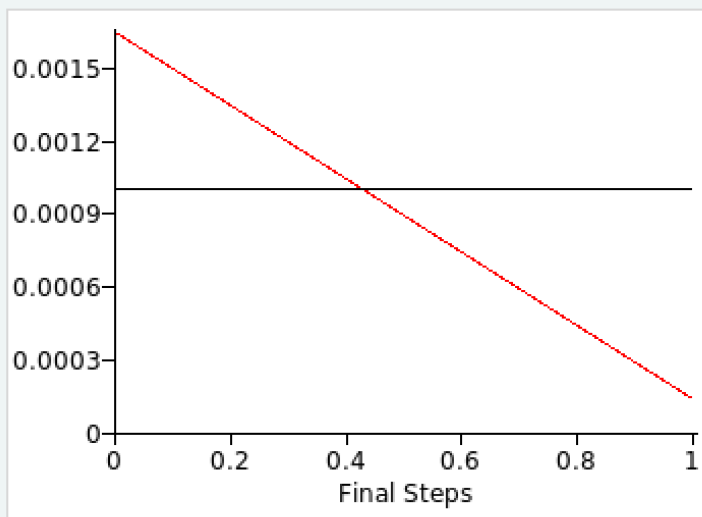
The building block of a DFT study: Automated Convergence to obtain accurate energy values

- Test for a range of energy cutoff and k-mesh values
- Convergence: 450 eV and 5x5x5 k-mesh
- Matches well with previous non-automated computational studies^[1]



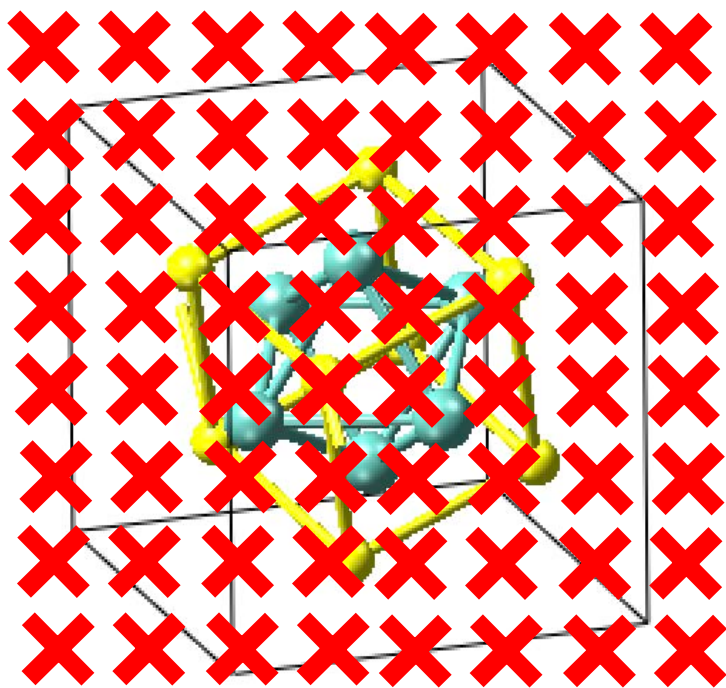
Data to display

Delta E/atom (eV)



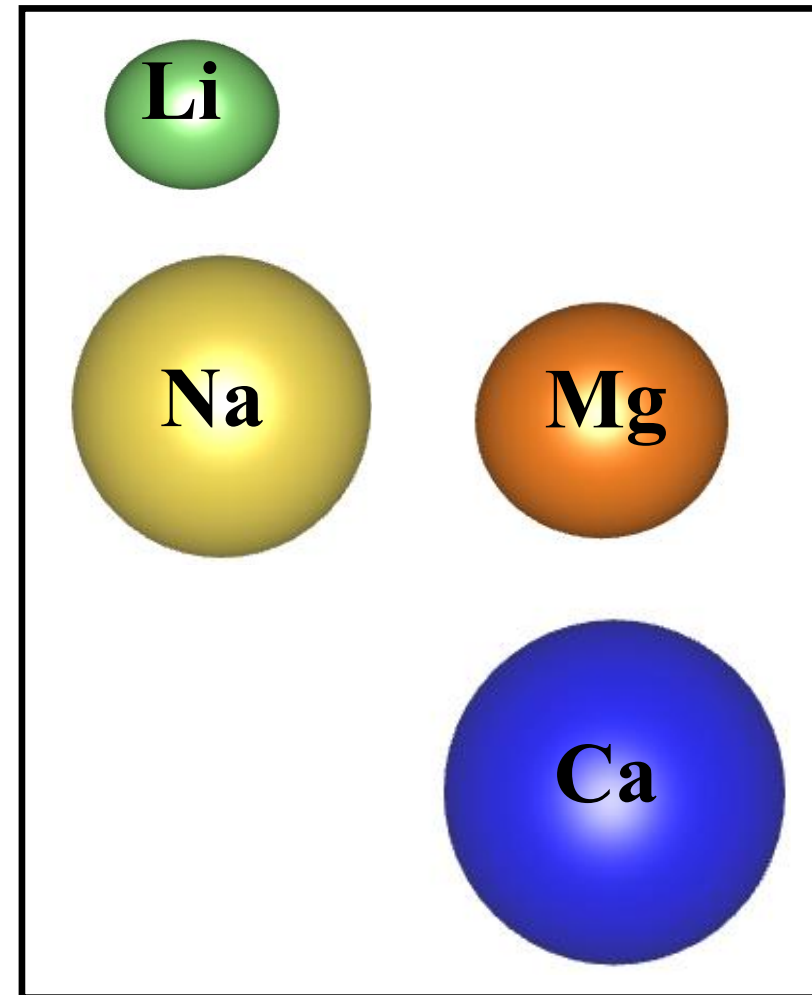
Intercalation site identification is a common problem with MVIBs

- Intercalation ion size differences lead to different intercalation sites
- Instead of waiting on costly NMR we can use modeling to identify sites



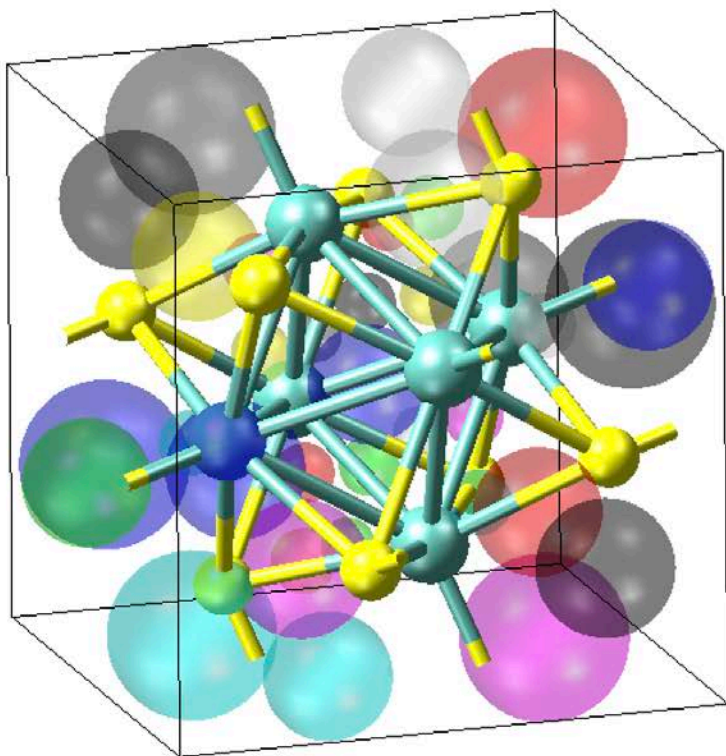
Mo₆S₈: Mo (teal), S (yellow)

← 100 Test Calculations



Automated intercalation site testing: Empty space finder

* (Mo₃S₄)₂ (P1) ~ 35358 minimized: CP_Spiedie

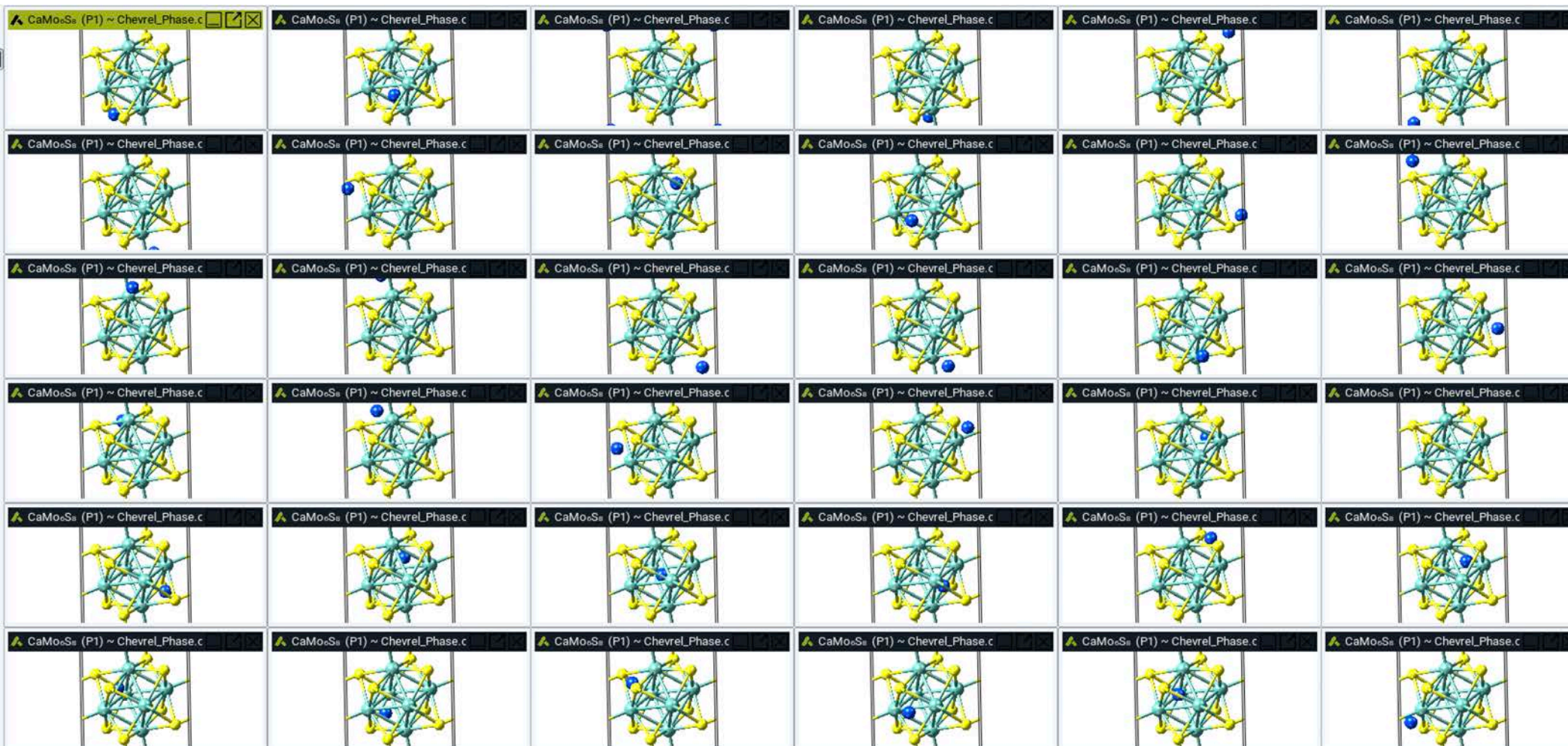


Id	Radius	Site	Coordination	X	Y	Z
4	0.345	1a	3	0.3984	0.5477	0.6625
5	0.376	1a	2	0.3444	0.3813	0.7749
6	0.376	1a	2	0.2250	0.6556	0.6185
7	0.292	1a	3	0.3675	0.3676	0.3676
8	0.345	1a	3	0.3375	0.6017	0.4522
9	0.292	1a	3	0.6325	0.6324	0.6324
10	0.308	1a	2	0.7042	0.8433	0.5229
11	0.345	1a	3	0.6625	0.3983	0.5478
12	0.683	1a	6	0.5000	0.5000	0.4999
13	0.345	1a	3	0.5477	0.6625	0.3983
14	0.376	1a	2	0.7750	0.3444	0.3815
15	0.345	1a	3	0.6016	0.4523	0.3375
16	0.376	1a	2	0.6556	0.6186	0.2251
17	0.973	1a	1	0.8867	0.7084	0.1362
18	0.801	1a	1	0.0731	0.5124	0.1288
19	0.974	1a	1	0.2917	0.8640	0.1130
20	0.376	1a	2	0.3815	0.7750	0.3444
21	0.801	1a	1	0.9269	0.4876	0.8712
22	0.376	1a	2	0.6185	0.2250	0.6556
23	0.974	1a	1	0.7083	0.1360	0.8870
24	0.972	1a	1	0.8637	0.1134	0.2919
25	0.742	1a	2	0.3357	0.9744	0.3729
26	0.800	1a	1	0.4879	0.8710	0.9268
27	0.972	1a	1	0.1363	0.8866	0.7081
28	0.742	1a	2	0.9743	0.3729	0.3357
29	0.743	1a	2	0.3730	0.3354	0.9744
30	0.743	1a	2	0.6270	0.6646	0.0256
31	0.742	1a	2	0.0257	0.6271	0.6643
32	0.742	1a	2	0.6643	0.0256	0.6271
33	0.801	1a	1	0.1288	0.0730	0.5123
34	0.801	1a	1	0.8712	0.9270	0.4877
35	0.800	1a	1	0.5121	0.1290	0.0732

- Reduce number of calculations from 100 to 35

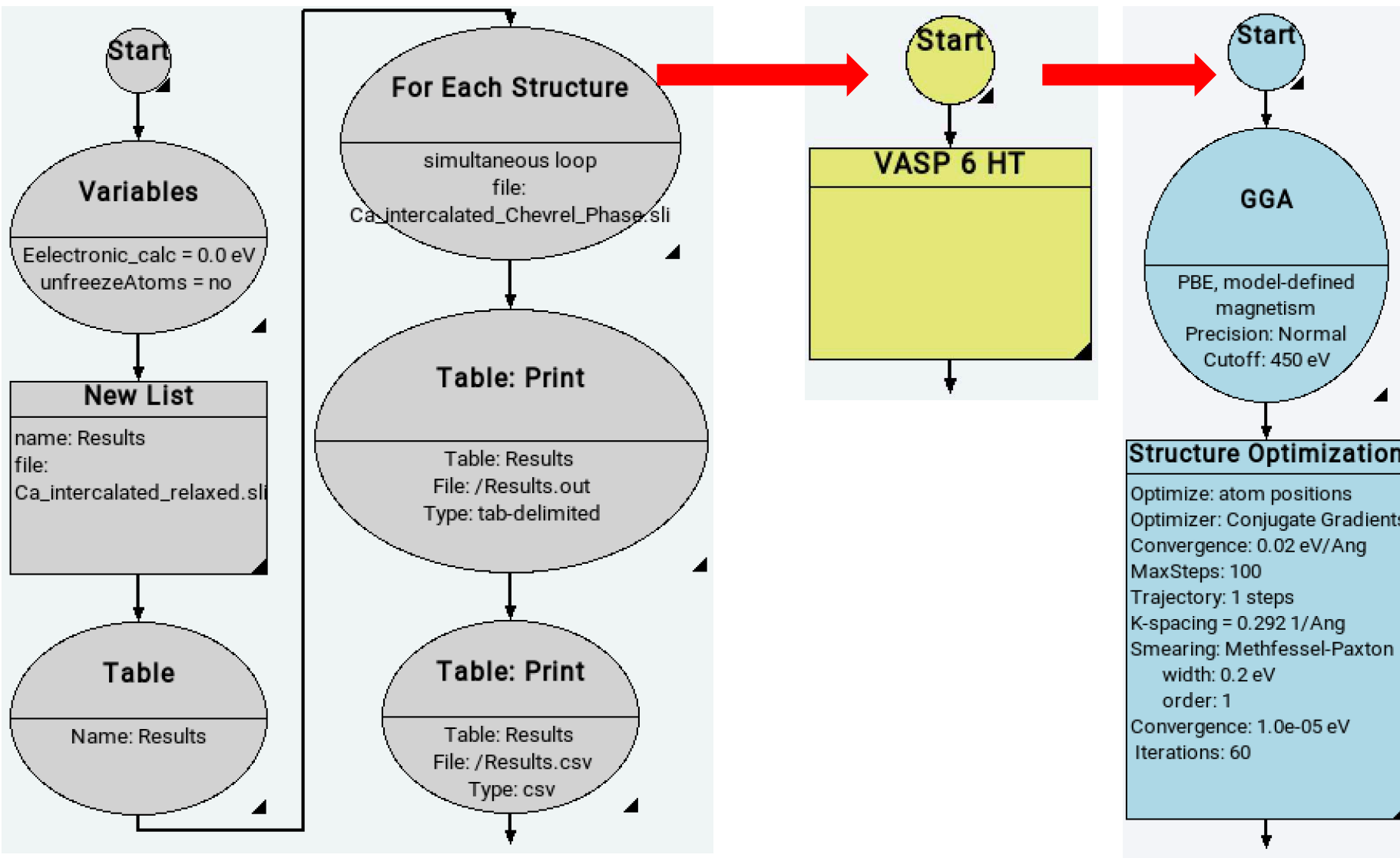
Minimum radius: Maximum radius:

Automated intercalation site testing: Move a Ca to each of the 35 possible intercalation sites



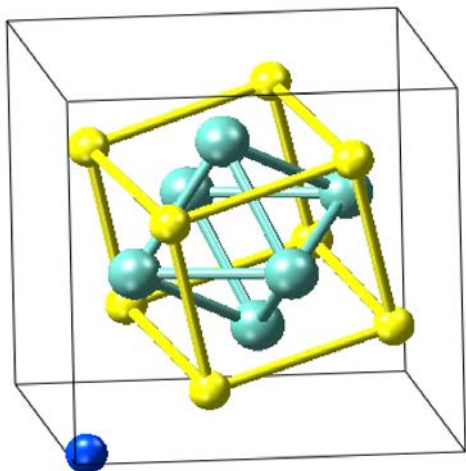
- Start with placing a Ca atom at a random position within the empty CP and moving it to each void
- Structure list containing 36 structures is created

Automated intercalation site testing: *MedeA VASP* high-throughput calculations

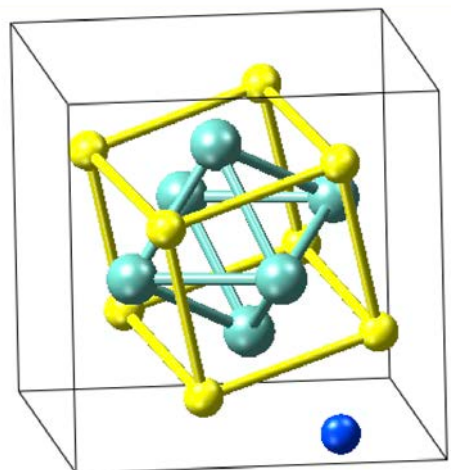


- You do not need know how to code to achieve fully automated complex multi-step calculations
- Pseudo coding is enough, and is conveniently guided by visualization tools

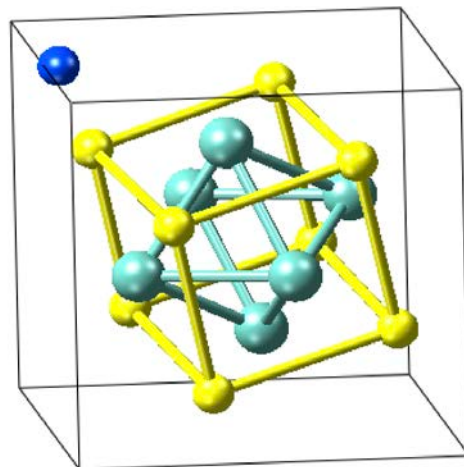
Automated intercalation site testing: sites identified



Cavity 1



Cavity 2



Cavity 3

Mg (blue), S (yellow), Mo (teal)

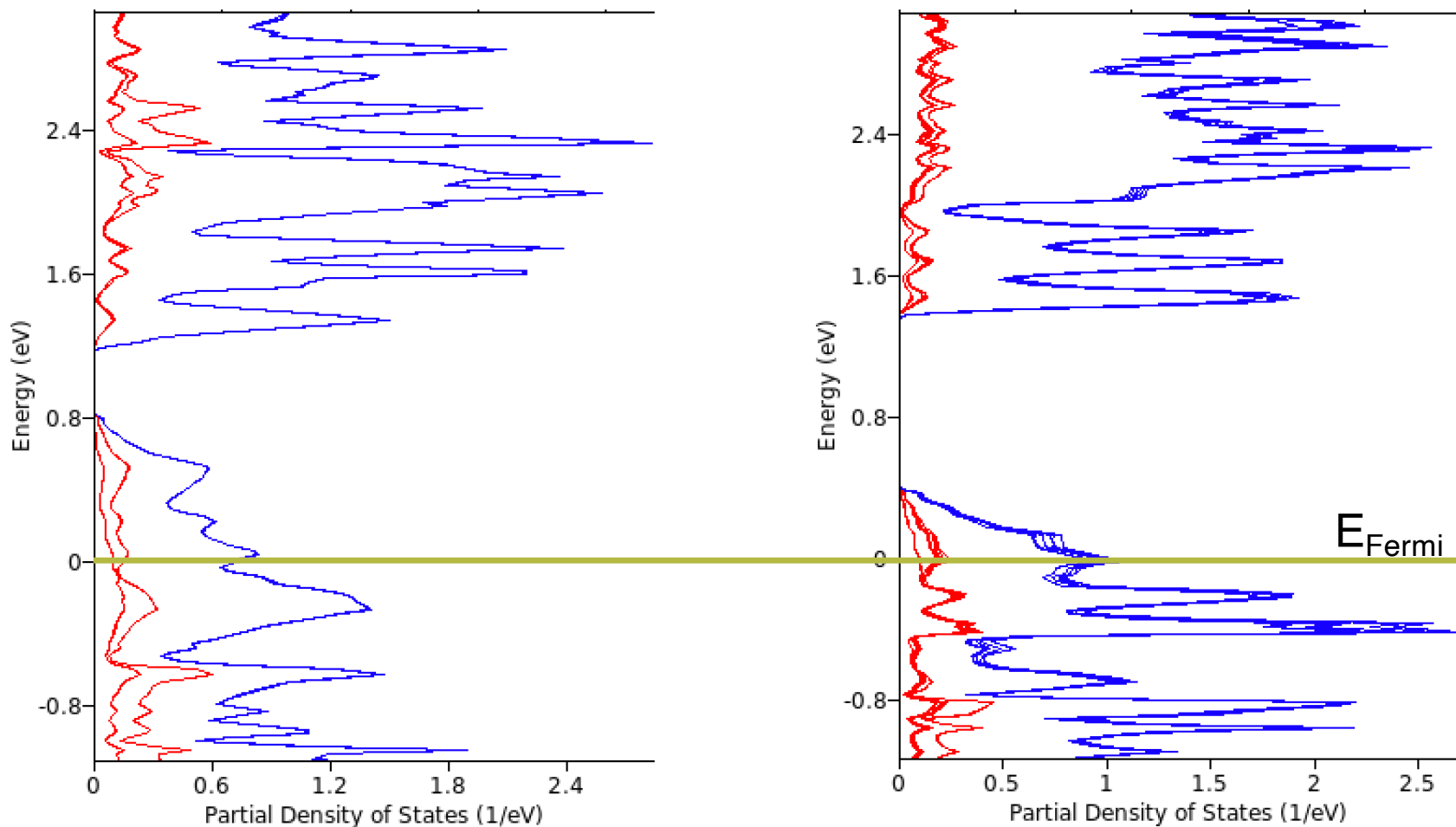
Metal	Cavities
Li	1, 2
Mg	1,2,3
Ca	1,3
Al	1,2,3

[1] *Journal of Power Sources*, Volume 306, 29 February 2016, Pages 431-436

[2] *ACS Appl. Energy Mater.* 2018, 1, 2, 440–446

[3] *Phys.Chem.Chem.Phys.*,2017,19,20684-20690 | 20685

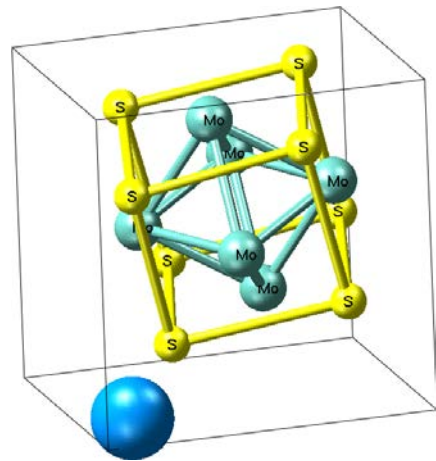
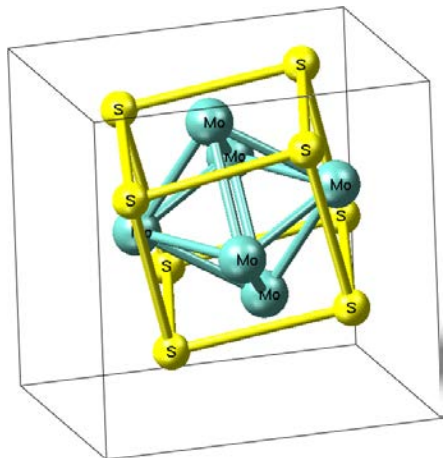
Electronic structure – without external visualization codes – DOS



Mo 4d (blue), S 3p (red)

- Finer k-mesh and type of smearing are automatically tailored to the DOS calculation
- Mo_6S_8 and CaMo_6S_8 are both metals
- Mo 4d-orbitals are hybridized with S 3p-orbitals near the Fermi level
- CaMo_6S_8 has higher conductivity
- Matches well with previous non-automated computational studies^[1]

Calculated Electrical Conductivity (σ_{elec})

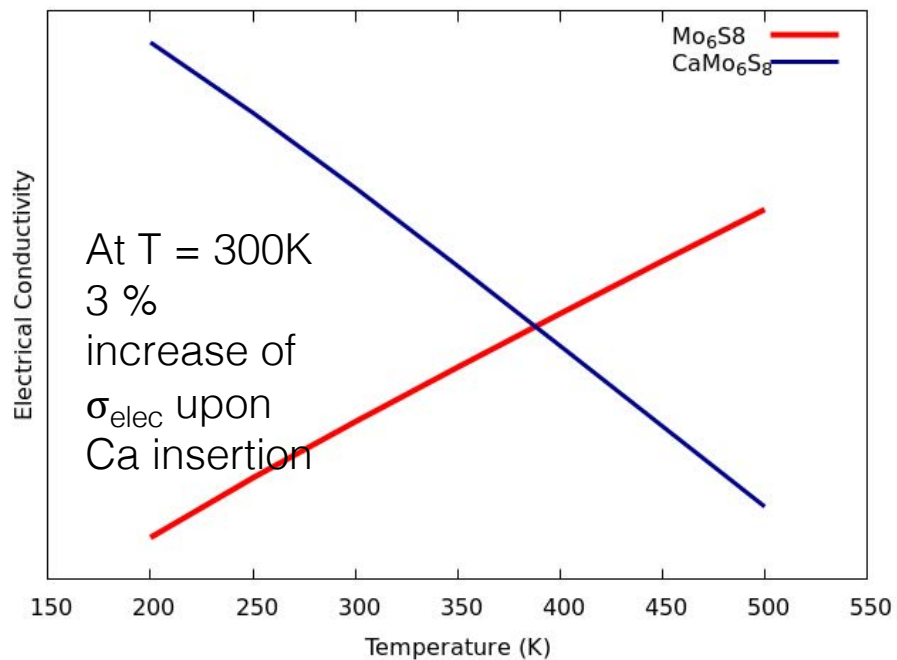


Band Structure

Electron Velocity and Effective Mass

Electronic Conductivity

- MedeA[®] Electronics
- Prediction of trends
- σ_{elec} of Mo_6S_8 increases with temperature whereas decreases in case of CaMo_6S_8
- At 300 K Ca insertion into Mo_6S_8 causes an increase of σ_{elec}
- In agreement with DOS, and gives a more in depth and temperature dependent analysis



Charge Analysis: Spatial distributions of electrons around each atom

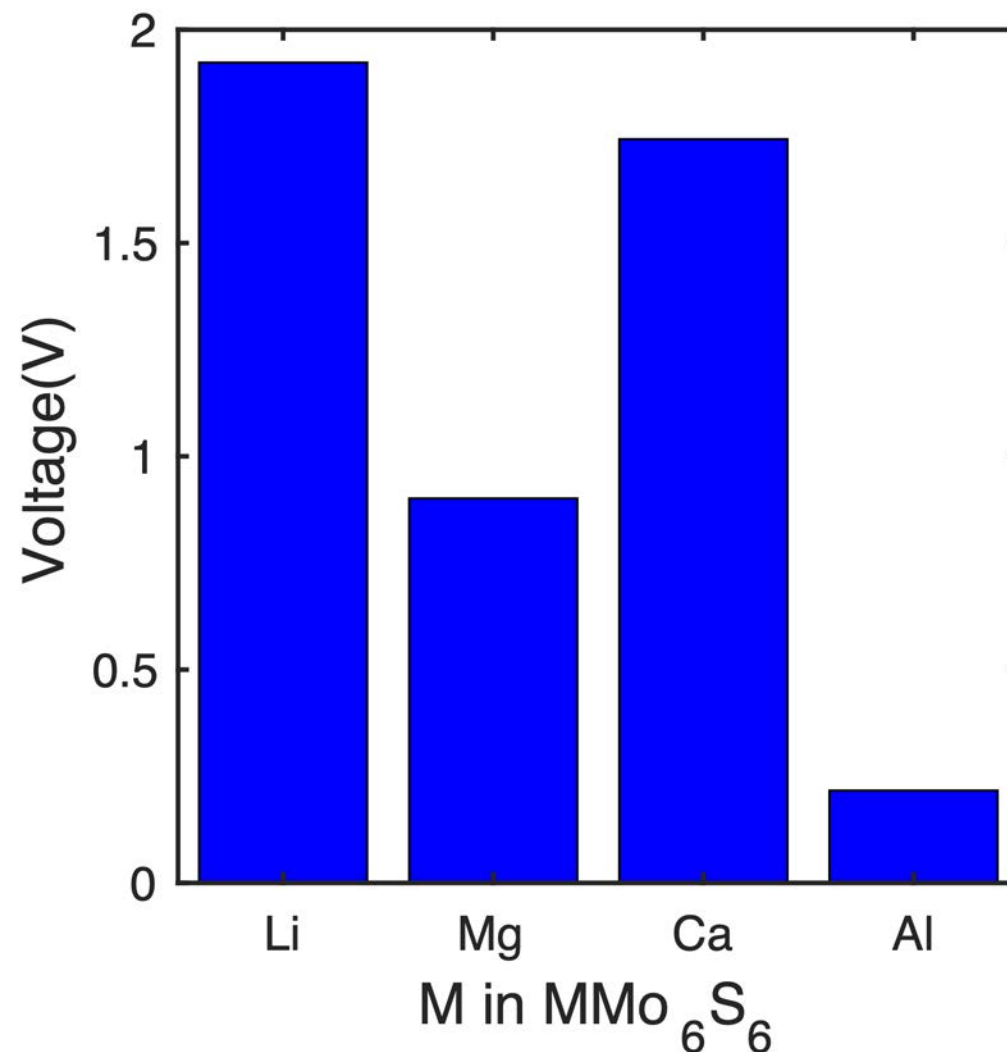
Atom	Mo6S8 Charge Difference (e)	CaMo6S8 Charge Difference (e)
Ca	-	1.49
Mo	0.99	0.93-0.95
S Type 1	-0.77	-0.85 - -0.94
S Type 2	-0.67	-0.87- -0.88

- MedeA[®] VASP Bader Charge Analysis
- Two types of S atoms
- The majority of the charge redistribution occurs on S atoms
- Matches well with previous non-automated computational studies^[1]

[1]
Phys.Chem.Chem.Phys.,2017,
19,20684-20690 | 20685

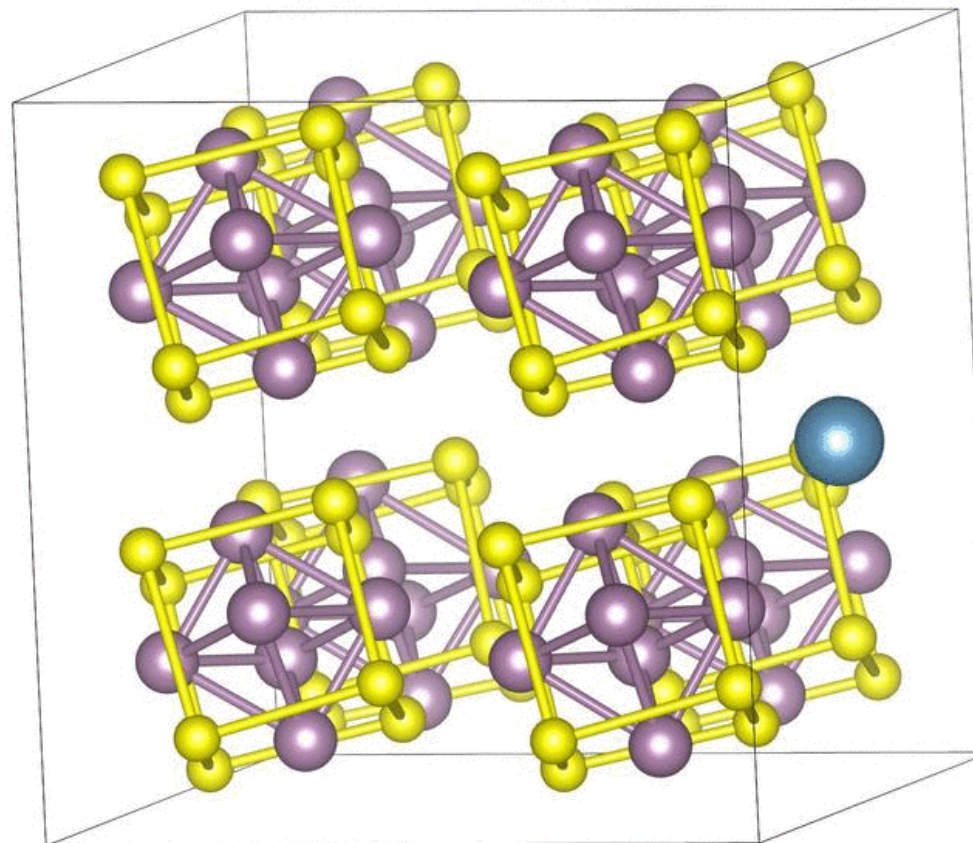
Average voltage for a $M\text{Mo}_6\text{S}_8$ unit cell

- Ca is significantly higher than Mg
- Matches well with previous non-automated computational studies^[1] and several experimental studies
- <1% volume change upon intercalation



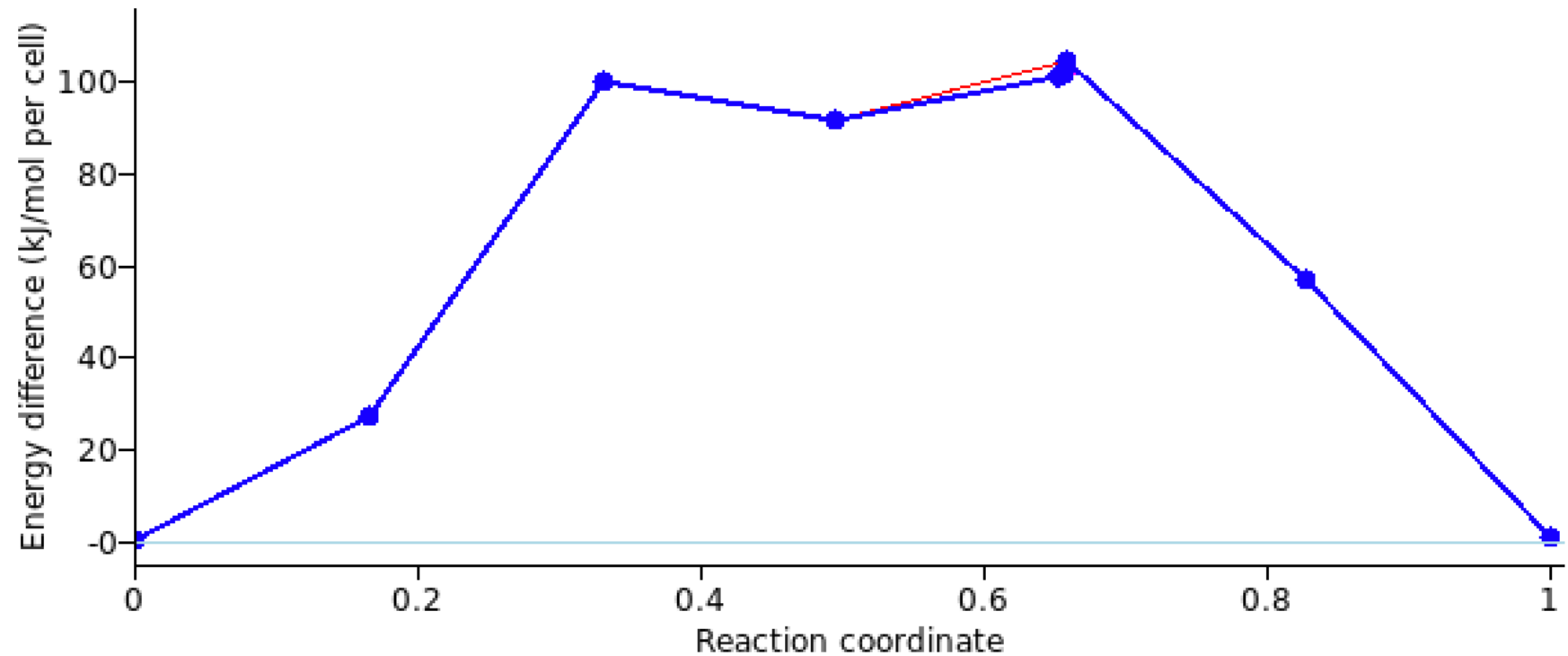
Diffusion kinetics – especially important with bulky MVIBs

- Cations experience an energy barrier as they transition from one intercalation site to the next
- MedeA[®] Transition State Search
 - Map reaction path and find transition states
 - Transition states in random directions
 - View images
 - Make movies

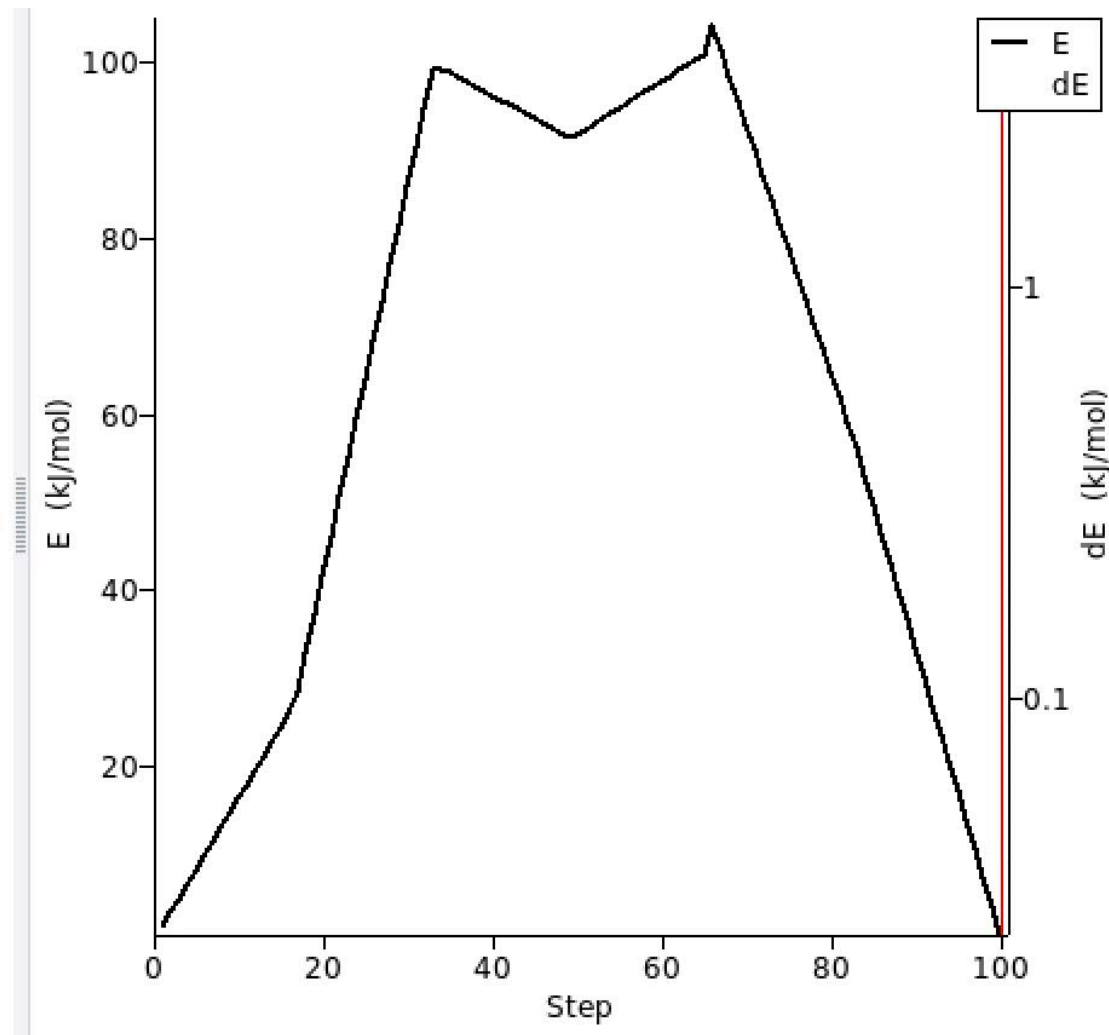
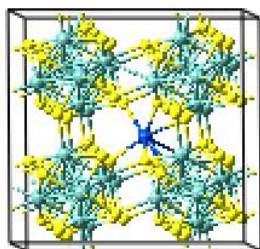
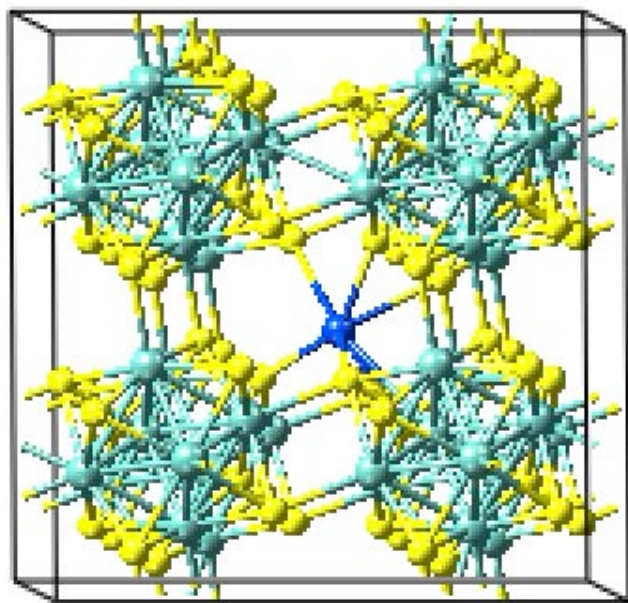


Energy barrier for Ca diffusion through a Mo₆S₈ supercell

- 1 eV energy barrier
- Matches well with previous computational studies^[1]



Energy barrier for Ca diffusion through a Mo_6S_8 supercell



Conclusion

The demand for next- generation batteries is at an all time high

We can design battery materials that cater to desired electrochemical properties






























Ca-ion batteries are a phenomenal contribution to next-generation batteries

MedeA is a comprehensive tool for investigations of next-generation batteries



materials design[®]

Acknowledgements

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 <p>Jessica Coll EMEA Academic Account Manager</p>	 <p>Brian Dron Commercial Director, Europe</p>	 <p>Volker Eyert Senior Scientist</p>	 <p>Laura Foster Financial Assistant in Finance & Operations</p>	 <p>Dave Rigby Senior Scientist</p>	 <p>Ken Roberts Head of Marketing Operations</p>	 <p>Xavier Rozanska Research Scientist</p>	 <p>Ray Shan Director of Support</p>	 <p>Walter Wolf Senior Scientist</p>	 <p>Marianna Yiannourakou Director of Product Management</p>	
 <p>Clive Freeman President & CEO</p>	 <p>Clint Geller Senior Advisor Scientist in Contract Research & Development</p>	 <p>Jörg-Rüdiger Hill Senior Software Engineer in Contract Research & Development</p>	 <p>Katherine Hollingsworth Marketing & Communications Manager</p>	 <p>Reynald Takchi Financial Assistant</p>	 <p>Siwen Wang Support & Application Scientist</p>	 <p>Erich Wimmer Chief Scientific Officer & Chairman of the Board</p>	 <p>Rene Windiks Senior Scientist, Battery Specialist</p>			

Key properties for batteries, that can be attained from atomistic modeling...

Materials Fabrication

- Free energies and phase diagrams: miscibility vs separation
- Elasticity: ductility, brittleness, hardness
- Permittivity Dielectric constants
- Piezoelectricity
- Diffusivity
- Thermal expansion
- Heat capacity
- Electronic structures

Cycling Behavior, Fast Charging

- Ion conductivity
- Electrical conductivity
- Thermal conductivity
- Electrochemical stability vs degradation • Phase transformation
- Volume change of particles

Interfaces of Electrodes, Electrolytes, Coatings, Binders, etc.

- Interphase morphology • Interfacial contact
- Current density
- Overpotentials
- Inter-diffusion & segregation
- Interfacial stabilities/delamination • Potential profiles

Diagnostics & Analysis

- XPS (core level shifts)
- NMR (chemical shifts, field gradients, paramagnetic shifts)
- Powder diffraction patterns • IR & Raman Spectra
- UV-Vis Spectra

Online Training and Demo

Li-Ion Batteries and Beyond: Driving Next-Generation Energy Storage with *MedeA*

Next Tuesday, January 26, 2021

USA/EUROPE:


10:00 am PST/1:00 pm EST

7:00 pm CEST

Register for the training:

<https://www.materialsdesign.com/webinar-register/li-ion-batteries-medea-jan-2021>

*Training open to everyone



Online Training and Demo:
Li-Ion Batteries and
Beyond: Driving Next-
Generation Energy
Storage with *MedeA*

Upcoming Webinar

Machine Learning Meets Quantum Chemistry: Using Theory, Data, and Experiments to Design Catalysts

Presented by Dr. Siwen Wang

February 2nd 3rd, 4th

Register www.materialsdesign.com/webinars



Li-Ion Batteries and Beyond

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 - <https://www.materialsdesign.com/analysis-tools>
 - [MedeA Environment](#)
 - [MedeA VASP](#)
 - [MedeA HT](#)
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- **For questions or comments contact**
 - Katherine Hollingsworth
 - khollingsworth@materialsdesign.com

Question and Answer Session



Dr. René Windiks

Materials Design



Dr. Taylor Juran

Materials Design

Questions about Materials Design Webinars

Katherine Hollingsworth

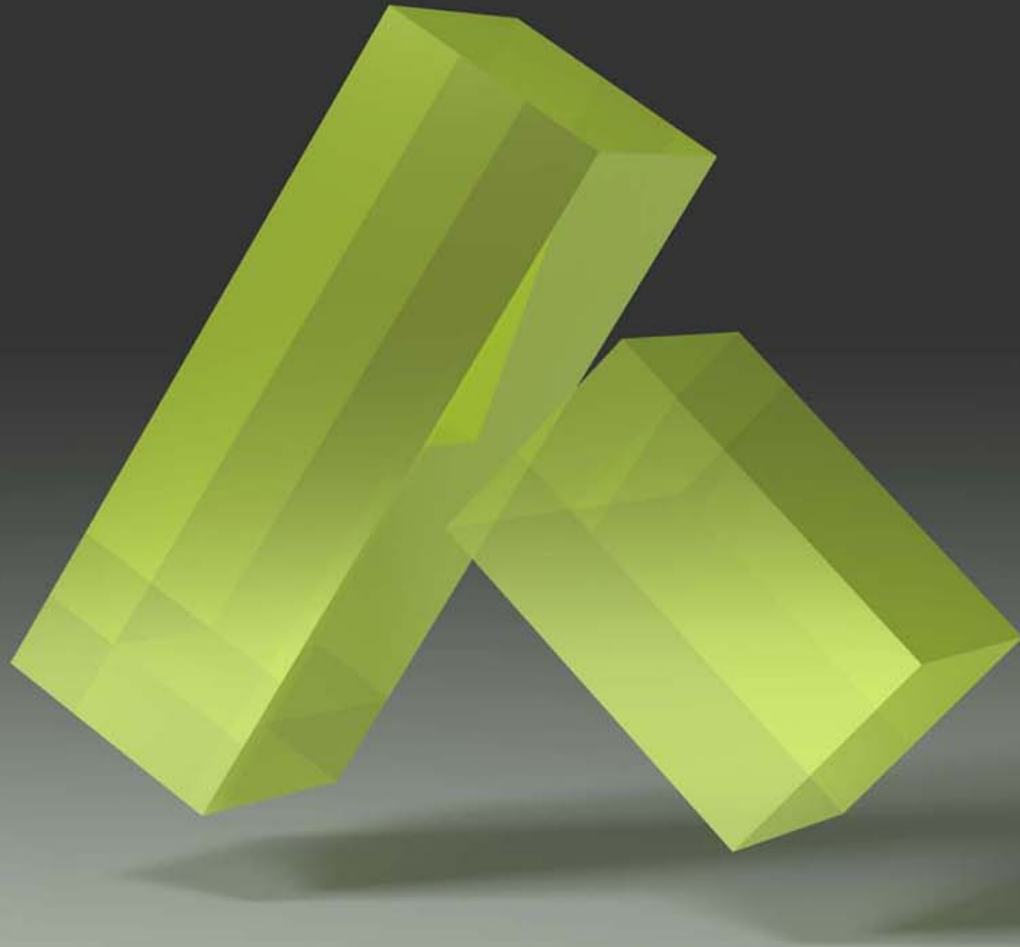
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